

LIS009196738B2

## (12) United States Patent

#### Sakata et al.

## (10) Patent No.: US 9,196,738 B2 (45) Date of Patent: Nov. 24, 2015

## (54) SEMICONDUCTOR DEVICE AND METHOD FOR MANUFACTURING THE SAME

(75) Inventors: Junichiro Sakata, Atsugi (JP);

Tetsunori Maruyama, Atsugi (JP); Yuki

Imoto, Atsugi (JP)

(73) Assignee: Semiconductor Energy Laboratory

Co., Ltd., Atsugi-shi, Kanagawa-ken

(JP)

(\*) Notice: Subject to any disclaimer, the term of this

patent is extended or adjusted under 35

U.S.C. 154(b) by 1123 days.

(21) Appl. No.: 12/965,197

(22) Filed: Dec. 10, 2010

(65) Prior Publication Data

US 2011/0140205 A1 Jun. 16, 2011

#### (30) Foreign Application Priority Data

Dec. 11, 2009 (JP) ...... 2009-281505

(51) Int. Cl.

**H01L 21/336** (2006.01)

**H01L 29/786** (2006.01)

(52) U.S. Cl.

CPC ...... *H01L 29/7869* (2013.01)

(58) Field of Classification Search

See application file for complete search history.

#### (56) References Cited

#### U.S. PATENT DOCUMENTS

5,731,856 A 3/1998 Kim et al. 5,744,864 A 4/1998 Cillessen et al.

6,294,274 B1 6,563,174 B2 6,727,522 B1 7,049,190 B2	5/2003 4/2004 5/2006	Kawazoe et al. Kawasaki et al. Kawasaki et al. Takeda et al.
7,061,014 B2 7,064,346 B2		Hosono et al. Kawasaki et al.
	(Con	tinued)

#### FOREIGN PATENT DOCUMENTS

EP 1737044 A 12/2006 EP 1770788 A 4/2007 (Continued)

#### OTHER PUBLICATIONS

International Search Report (Application No. PCT/JP2010/072595) Dated Feb. 1, 2011.

(Continued)

Primary Examiner — Colleen A Matthews

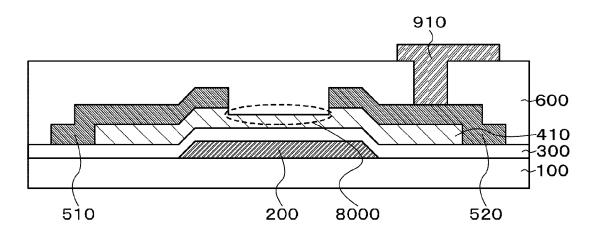
Assistant Examiner — Colleen E Snow

(74) Attorney, Agent, or Firm — Eric J. Robinson; Robinson
Intellectual Property Law Office, P.C.

#### (57) ABSTRACT

Many of the physical properties of a silicon semiconductor have already been understood, whereas many of the physical properties of an oxide semiconductor have been still unclear. In particular, an adverse effect of an impurity on an oxide semiconductor has been still unclear. In view of the above, a structure is disclosed in which an impurity that influences electrical characteristics of a semiconductor device including an oxide semiconductor layer is prevented or is eliminated. A semiconductor device which includes a gate electrode, an oxide semiconductor layer, and a gate insulating layer provided between the gate electrode and the oxide semiconductor layer and in which the nitrogen concentration in the oxide semiconductor layer is  $1 \times 10^{20}$  atoms/cm<sup>3</sup> or less is provided.

#### 16 Claims, 16 Drawing Sheets



# US 9,196,738 B2 Page 2

(56)	Referen	nces Cited		/0038929 A1		Chang
U.S.	PATENT	DOCUMENTS	2008	/0050595 A1 /0073653 A1	3/2008	Nakagawara et al. Iwasaki
7,105,868 B2	0/2006	Nause et al.		/0083950 A1 /0106191 A1		Pan et al. Kawase
7,103,808 B2 7,211,825 B2		Shih et al	2008	/0128689 A1	6/2008	Lee et al.
7,262,463 B2		Hoffman		/0129195 A1 /0166834 A1		Ishizaki et al. Kim et al.
7,282,782 B2 7,297,977 B2		Hoffman et al. Hoffman et al.		/0182358 A1		Cowdery-Corvan et al.
7,323,356 B2		Hosono et al.		/0224133 A1	9/2008	Park et al.
7,368,067 B2		Yata et al.				Hoffman et al. Ito et al.
7,385,224 B2 7,402,506 B2		Ishii et al. Levy et al.				Lee et al.
7,411,209 B2	8/2008	Endo et al.				Park et al.
7,453,065 B2		Saito et al. Iwasaki				Kim et al. Ryu et al.
7,453,087 B2 7,462,862 B2		Hoffman et al.		/0045397 A1	2/2009	Iwasaki
7,468,304 B2		Kaji et al.		/0068773 A1 /0073325 A1		Lai et al. Kuwabara et al.
7,501,293 B2 7,564,055 B2		Ito et al. Hoffman		/00/3323 A1 /0090914 A1		Yano et al.
7,674,650 B2		Akimoto et al.	2009	/0114910 A1	5/2009	Chang
7,732,819 B2		Akimoto et al.		/0134399 A1 /0152541 A1		Sakakura et al. Maekawa et al.
7,791,074 B2 7,935,582 B2		Iwasaki Iwasaki				Hosono et al.
7,956,361 B2		Iwasaki				Hosono et al.
8,154,024 B2		Iwasaki		/0084648 A1 /0123132 A1*		Watanabe Nakata et al 257/43
8,188,467 B2 8,207,063 B2		Itagaki et al. Cowdery-Corvan et al.		/0276689 A1 1	1/2010	Iwasaki
8,735,229 B2*	5/2014	Son et al 438/149				Iwasaki Wana at al
8,785,240 B2 8,962,457 B2		Watanabe Watanabe	2013/	/0221348 A1	8/2013	Yano et al.
2001/0046027 A1		Tai et al.		FOREIGN	N PATE	NT DOCUMENTS
2002/0056838 A1		Ogawa				
2002/0132454 A1 2003/0189401 A1		Ohtsu et al. Kido et al.	EP EP		177 A 347 A	7/2008 9/2010
2003/0218222 A1		Wager et al.	EP		539 A	6/2011
2004/0038446 A1		Takeda et al.	JP	60-1988		10/1985
2004/0127038 A1 2005/0017244 A1		Carcia et al. Hoffman et al.	JP JP	63-2100 63-2100		8/1988 8/1988
2005/0017302 A1	1/2005	Hoffman	JР	63-2100		8/1988
2005/0199959 A1		Chiang et al. Carcia et al.	JP	63-2155		9/1988
2006/0035452 A1 2006/0043377 A1		Hoffman et al.	JP JP	63-2391 63-2658		10/1988 11/1988
2006/0091793 A1	5/2006	Baude et al.	JP	05-2517	705 A	9/1993
2006/0108529 A1 2006/0108636 A1		Saito et al. Sano et al.	JP JP	08-2647		10/1996
2006/0110867 A1		Yabuta et al.	JP	11-5053 2000-0442		5/1999 2/2000
2006/0113536 A1		Kumomi et al.	JP	2000-1509		5/2000
2006/0113539 A1 2006/0113549 A1		Sano et al. Den et al.	JP JP	2002-0763 2002-2898		3/2002 10/2002
2006/0113565 A1	6/2006	Abe et al.	JP	2003-0860		3/2003
2006/0169973 A1 2006/0170111 A1		Isa et al. Isa et al.	JP	2003-0868		3/2003
2006/0170111 A1 2006/0197092 A1		Hoffman et al.	JP JP	2004-1039 2004-2736		4/2004 9/2004
2006/0208977 A1		Kimura	JP	2004-2737		9/2004
2006/0228974 A1 2006/0231882 A1		Thelss et al. Kim et al.	JP JP	2005-2170 2007-1039		8/2005 4/2007
2006/0238135 A1	10/2006	Kimura	JP	2007-1039		5/2007
2006/0244107 A1		Sugihara et al.	JP	2007-5192		7/2007
2006/0284171 A1 2006/0284172 A1	12/2006	Levy et al. Ishii	JP JP	2007-3114 2008-2819		11/2007 11/2008
2006/0292777 A1	12/2006	Dunbar	JP	2008-2830		11/2008
2007/0024187 A1 2007/0046191 A1	2/2007 3/2007	Shin et al.	JP	2008-3005		12/2008
2007/0040191 A1 2007/0052025 A1		Yabuta	JP WO	2009-1412 WO-2004/1143		6/2009 12/2004
2007/0054507 A1		Kaji et al.	WO	WO-2005/0156	543	2/2005
2007/0072439 A1 2007/0090365 A1		Akimoto et al. Hayashi et al.	WO WO	WO-2005/0740 WO-2007/0298		8/2005 3/2007
2007/0108446 A1		Akimoto	wo	WO-2007/02582		5/2007
2007/0152217 A1		Lai et al.	WO	WO-2008/0915	504	7/2008
2007/0172591 A1 2007/0187678 A1		Seo et al. Hirao et al.	WO WO	WO-2008/1268 WO-2008/1430		10/2008 11/2008
2007/0187760 A1	8/2007	Furuta et al.	wo	WO-2008/1497		12/2008
2007/0194379 A1		Hosono et al.		OTH	FR PIT	BLICATIONS
2007/0252928 A1 2007/0272922 A1		Ito et al. Kim et al.		OIII	LICI U	
2007/0287296 A1	12/2007				ation No	o. PCT/JP2010/072595) Dated Feb.
2008/0006877 A1		Mardilovich et al.	1, 201		י מון	III-1 M-171- 7 O FT : E
2008/0038882 A1	2/2008	Takechi et al.	rortun	ato.E et al., "Wid	ie-Bandg	gap High-Mobility ZnO Thin-Film

#### (56) References Cited

#### OTHER PUBLICATIONS

Transistors Produced at Room Temperature,", Appl. Phys. Lett. (Applied Physics Letters), Sep. 27, 2004, vol. 85, No. 13, pp. 2541-2543

Dembo.H et al., "RFCPUS on Glass and Plastic Substrates Fabricated by TFT Transfer Technology,", IEDM 05: Technical Digest of International Electron Devices Meeting, Dec. 5, 2005, pp. 1067-1069.

Ikeda.T et al., "Full-Functional System Liquid Crystal Display Using CG-Silicon Technology,", SID Digest '04: SID International Symposium Digest of Technical Papers, 2004, vol. 35, pp. 860-863.

Nomura.K et al., "Room-Temperature Fabrication of Transparent Flexible Thin-Film Transistors using Amorphous Oxide Semiconductors,", Nature, Nov. 25, 2004, vol. 432, pp. 488-492.

Park.J et al., "Improvements in the Device Characteristics of Amorphous Indium Gallium Zinc Oxide Thin-Film Transistors by Ar Plasma Treatment,", Appl. Phys. Lett. (Applied Physics Letters), Jun. 26, 2007, vol. 90, No. 26, pp. 262106-1-262106-3.

Takahashi.M et al., "Theoretical Analysis of IGZO Transparent Amorphous Oxide Semiconductor,", IDW '08: Proceedings of the 15th International Display Workshops, Dec. 3, 2008, pp. 1637-1640. Hayashi.R et al., "42.1: Invited Paper: Improved Amorphous In—Ga—Zn—O TFTs,", SID Digest '08: SID International Symposium Digest of Technical Papers, May 20, 2008, vol. 39, pp. 621-624.

Prins.M et al., "A Ferroelectric Transparent Thin-Film Transistor,", Appl. Phys. Lett. (Applied Physics Letters), Jun. 17, 1996, vol. 68, No. 25, pp. 3650-3652.

Nakamura.M et al., "The phase relations in the In2O3—Ga2ZnO4—ZnO system at 1350° C.,", Journal of Solid State Chemistry, Aug. 1, 1991, vol. 93, No. 2, pp. 298-315.

Kimizuka.N et al., "Syntheses and Single-Crystal Data of Homologous Compounds, In2O3(ZnO)m (m=3,4, and 5), InGaO3(ZnO)3, and Ga2O3(ZnO)m (m=7,8,9, and 16) in the In2O3—ZnGa2O4—ZnO System,", Journal of Solid State Chemistry, Apr. 1, 1995, vol. 116, No. 1, pp. 170-178.

Nomura.K et al., "Thin-Film Transistor Fabricated in Single-Crystalline Transparent Oxide Semiconductor,", Science, May 23, 2003, vol. 300, No. 5623, pp. 1269-1272.

Masuda.S et al., "Transparent thin film transistors using ZnO as an active channel layer and their electrical properties,", J. Appl. Phys. (Journal of Applied Physics), Feb. 1, 2003, vol. 93, No. 3, pp. 1624-1630

Asakuma.N et al., "Crystallization and Reduction of Sol-Gel-Derived Zinc Oxide Films by Irradiation with Ultraviolet Lamp,", Journal of Sol-Gel Science and Technology, 2003, vol. 26, pp. 181-184. Osada.T et al., "15.2: Development of Driver-Integrated Panel using Amorphous In—Ga—Zn—Oxide TFT,", SID Digest '09: SID International Symposium Digest of Technical Papers, May 31, 2009, pp. 184-187

Nomura.K et al., "Carrier transport in transparent oxide semiconductor with intrinsic structural randomness probed using single-crystal-line InGaO3(ZnO)5 films,", Appl. Phys. Lett. (Applied Physics Letters), Sep. 13, 2004, vol. 85, No. 11, pp. 1993-1995.

Li.C et al., "Modulated Structures of Homologous Compounds InMO3(ZnO)m (M=In,Ga; m=Integer) Described by Four-Dimensional Superspace Group,", Journal of Solid State Chemistry, 1998, vol. 39, pp. 347-355.

Son.K et al., "42.4L: Late-News Paper: 4 Inch QVGA AMOLED Driven by the Threshold Voltage Controlled Amorphous GIZO (Ga2O3—In2O3—ZnO) TFT,", SID Digest '08: SID International Symposium Digest of Technical Papers, May 20, 2008, vol. 39, pp. 633-636.

Lee.J et al., "World's Largest (15-Inch) XGA AMLCD Panel Using IGZO Oxide TFT,", SID Digest '08: SID International Symposium Digest of Technical Papers, May 20, 2008, vol. 39, pp. 625-628.

Nowatari.H et al., "60.2: Intermediate Connector With Suppressed Voltage Loss for White Tandem OLEDs,", SID Digest '09: SID International Symposium Digest of Technical Papers, May 31, 2009, vol. 40, pp. 899-902.

Kanno.H et al., "White Stacked Electrophosphorecent Organic Light-Emitting Devices Employing MOO3 as a Charge-Generation Layer,", Adv. Mater. (Advanced Materials), 2006, vol. 18, No. 3, pp. 339-342.

Tsuda.K et al., "Ultra Low Power Consumption Technologies for Mobile TFT-LCDs,", IDW '02: Proceedings of the 9th International Display Workshops, Dec. 4, 2002, pp. 295-298.

Walle.C, "Hydrogen as a Cause of Doping in Zinc Oxide,", Phys. Rev. Lett. (Physical Review Letters), Jul. 31, 2000, vol. 85, No. 5, pp. 1012-1015.

Fung.T et al., "2-D Numerical Simulation of High Performance Amorphous In—Ga—Zn—O TFTs for Flat Panel Displays,", AM-FPD '08 Digest of Technical Papers, Jul. 2, 2008, pp. 251-252, The Japan Society of Applied Physics.

Jeong J et al., "3.1: Distinguished Paper: 12.1-Inch WXGA AMOLED Display Driven by Indium—Gallium—Zinc Oxide TFTs Array,", SID Digest '08: SID International Symposium Digest of Technical Papers, May 20, 2008, vol. 39, No. 1, pp. 1-4.

Park J et al., "High performance amorphous oxide thin film transistors with self-aligned top-gate structure,", IEDM 09: Technical Digest of International Electron Devices Meeting, Dec. 7, 2009, pp. 191-194.

Kurokawa.Y et al., "UHF RFCPUS on Flexible and Glass Substrates for Secure RFID Systems,", Journal of Solid-State Circuits, 2008, vol. 43, No. 1, pp. 292-299.

Ohara.H et al., "Amorphous In—Ga—Zn—Oxide TFTs with Suppressed Variation for 4.0 inch QVGA AMOLED Display,", AM-FPD '09 Digest of Technical Papers, Jul. 1, 2009, pp. 227-230, The Japan Society of Applied Physics.

Coates.D et al., "Optical Studies of the Amorphous Liquid-Cholesteric Liquid Crystal Transition:The "Blue Phase",", Physics Letters, Sep. 10, 1973, vol 45A, No. 2, pp. 115-116.

Cho.D et al., "21.2:Al and Sn-Doped Zinc Indium Oxide Thin Film Transistors for AMOLED Back-Plane,", SID Digest '09: SID International Symposium Digest of Technical Papers, May 31, 2009, pp. 280-283.

Lee.M et al., "15.4:Excellent Performance of Indium—Oxide-Based Thin-Film Transistors by DC Sputtering,", SID Digest '09: SID International Symposium Digest of Technical Papers, May 31, 2009, pp. 191-193.

Jin.D et al., "65.2:Distinguished Paper: World-Largest (6.5") Flexible Full Color Top Emission AMOLED Display on Plastic Film and its Bending Properties,", SID Digest '09: SID International Symposium Digest of Technical Papers, May 31, 2009, pp. 983-985.

Sakata. J et al., "Development of 4.0-In. AMOLED Display With Driver Circuit Using Amorphous In—Ga—Zn—Oxide TFTs,", IDW '09: Proceedings of the 16th International Display Workshops, 2009, pp. 689-692.

Park J et al., "Amorphous Indium—Gallium—Zinc Oxide TFTs and Their Application for Large Size AMOLED,", AM-FPD '08 Digest of Technical Papers, Jul. 2, 2008, pp. 275-278.

Park.S et al., "Challenge to Future Displays: Transparent AM-OLED Driven by Peald Grown ZnO TFT,", IMID '07 Digest, 2007, pp. 1249-1252.

Godo.H et al., "Temperature Dependence of Characteristics and Electronic Structure For Amorphous In—Ga—Zn—Oxide TFT,", AM-FPD '09 Digest of Technical Papers, Jul. 1, 2009, pp. 41-44.

Osada.T et al., "Development of Driver-Integrated Panel Using Amorphous In—Ga—Zn—Oxide TFT,", AM-FPD '09 Digest of Technical Papers, Jul. 1, 2009, pp. 33-36.

Hirao.T et al., "Novel Top-Gate Zinc Oxide Thin-Film Transistors (ZnO TFTs) for AMLCDS,", Journal of the SID, 2007, vol. 15, No. 1, pp. 17-22.

Hosono.H, "68.3:Invited Paper:Transparent Amorphous Oxide Semiconductors for High Performance TFT,", SID Digest '07: SID International Symposium Digest of Technical Papers, 2007, vol. 38, pp. 1830-1833.

Godo.H et al., "P-9:Numerical Analysis on Temperature Dependence of Characteristics of Amorphous In—Ga—Zn—Oxide TFT,", SID Digest of Technical Papers, May 31, 2009, pp. 1110-1112.

Ohara.H et al., "21.3:4.0 In. QVGA AMOLED Display Using In—Ga—Zn—Oxide TFTs With a Novel Passivation Layer,", SID

#### (56) References Cited

#### OTHER PUBLICATIONS

Digest '09: SID International Symposium Digest of Technical Papers, May 31, 2009, pp. 284-287.

Miyasaka M, "SUFTLA Flexible Microelectronics on Their Way to Business,", SID Digest '07: SID International Symposium Digest of Technical Papers, 2007, vol. 38, pp. 1673-1676.

Chern.H et al., "An Analytical Model for the Above-Threshold Characteristics of Polysilicon Thin-Film Transistors,", IEEE Transactions on Electron Devices, Jul. 1, 1995, vol. 42, No. 7, pp. 1240-1246.

Kikuchi.H et al., "39.1:Invited Paper:Optically Isotropic Nano-Structured Liquid Crystal Composites For Display Applications,", SID Digest '09: SID International Symposium Digest of Technical Papers, May 31, 2009, pp. 578-581.

Asaoka.Y et al., "29.1:Polarizer-Free Reflective LCD Combined With Ultra Low-Power Driving Technology,", SID Digest '09: SID International Symposium Digest of Technical Papers, May 31, 2009, pp. 395-398.

Lee.H et al., "Current Status of, Challenges to, and Perspective View of AM-OLED,", IDW '06: Proceedings of the 13th International Display Workshops, Dec. 7, 2006, pp. 663-666.

Kikuchi. H et al., "62.2:Invited Paper: Fast Electro-Optical Switching in Polymer-Stabilized Liquid Crystalline Blue Phases For Display Application,", SID Digest '07: SID International Symposium Digest of Technical Papers, 2007, vol. 38, pp. 1737-1740.

中村真佐樹, "Synthesis of Homologous Compound with New Long-Period Structure,", NIRIM Newsletter, Mar. 1, 1995, vol. 150, pp.1-4. Kikuchi.H et al., "Polymer-Stabilized Liquid Crystal Blue Phases,", Nature Materials, Sep. 1, 2002, vol. 1, pp. 64-68.

Kimizuka.N. et al., "Spinel,YBFE2O4, and YB2FE3O7 Types of Structures for Compounds in the In2O3 and SC2O3—A2O3—BO Systems [A; Fe, Ga, or Al; B: Mg, Mn, Fe, Ni, Cu, or Zn] at Temperatures over 1000° C.,", Journal of Solid State Chemistry, 1985, vol. 60, pp. 382-384.

Kitzerow.H et al., "Observation of Blue Phases in Chiral Networks,", Liquid Crystals, 1993, vol. 14, No. 3, pp. 911-916.

Costello.M et al., "Electron Microscopy of a Cholesteric Liquid Crystal and Its Blue Phase,", Phys. Rev. A (Physical Review. A), May 1, 1984, vol. 29, No. 5, pp. 2957-2959.

Meiboom.S et al., "Theory of the Blue Phase of Cholesteric Liquid Crystals,", Phys. Rev. Lett. (Physical Review Letters), May 4, 1981, vol. 46, No. 18, pp. 1216-1219.

Park.Sang-Hee et al., "42.3: Transparent ZnO Thin Film Transistor for the Application of High Aperture Ratio Bottom Emission AM-OLED Display,", SID DIGEST '08: SID International Symposium Digest of Technical Papers, May 20, 2008, vol. 39, pp. 629-632.

Orita.M et al., "Mechanism of Electrical Conductivity of Transparent InGaZnO4,", Phys. Rev. B (Physical Review. B), Jan. 15, 2000, vol. 61, No. 3, pp. 1811-1816.

Nomura.K et al., "Amorphous Oxide Semiconductors for High-Performance Flexible Thin-Film Transistors,", Jpn. J. Appl. Phys. (Japanese Journal of Applied Physics), 2006, vol. 45, No. 5B, pp. 4303-4308.

Janotti. A et al., "Native Point Defects in ZnO,", Phys. Rev. B (Physical Review. B), 2007, vol. 76, pp. 165202-1-165202-22.

Park.J et al., "Electronic Transport Properties of Amorphous Indium—Gallium—Zinc Oxide Semiconductor Upon Exposure to Water,", Appl. Phys. Lett. Applied Physics Letters), 2008, vol. 92, pp. 072104-1-072104-3.

Hsieh.H et al., "P-29:Modeling of Amorphous Oxide Semiconductor Thin Film Transistors and Subgap Density of States,", SID Digest '08: SID International Symposium Digest of Technical Papers, 2008, vol. 39, pp. 1277-1280.

Janotti.A et al., "Oxygen Vacancies in ZnO,", Appl. Phys. Lett. (Applied Physics Letters), 2005, vol. 87, pp. 122102-1-122102-3. Oba.F et al., "Defect energetics in ZnO: A hybrid Hartree-Fock density functional study,", Phys. Rev. B (Physical Review. B), 2008, vol. 77, pp. 245202-1-245202-6.

Orita.M et al., "Amorphous transparent conductive oxide InGaO3(ZnO)m (m<4):a Zn4s conductor,", Philosophical Magazine, 2001, vol. 81, No. 5, pp. 501-515.

Hosono. H et al., "Working hypothesis to explore novel wide band gap electrically conducting amorphous oxides and examples,", J. Non-Cryst. Solids (Journal of Non-Crystalline Solids), 1996, vol. 198-200, pp. 165-169.

Mo.Y et al., "Amorphous Oxide TFT Backplanes for Large Size AMOLED Displays,", IDW '08: Proceedings of the 6th International Display Workshops, Dec. 3, 2008, pp. 581-584.

Kim.S et al., "High-Performance oxide thin film transistors passivated by various gas plasmas,", 214th ECS Meeting, 2008, No. 2317.

Clark.S et al., "First Principles Methods Using CASTEP,", Zeitschrift für Kristallographie, 2005, vol. 220, pp. 567-570.

Lany.S et al., "Dopability, Intrinsic Conductivity, and Nonstoichiometry of Transparent Conducting Oxides,", Phys. Rev. Lett. (Physical Review Letters), Jan. 26, 2007, vol. 98, pp. 045501-1-045501-4.

Park.J et al., "Dry etching of ZnO films and plasma-induced damage to optical properties,", J. Vac. Sci. Technol. B (Journal of Vacuum Science & Technology B), Mar. 1, 2003, vol. 21, No. 2, pp. 800-803. Oh.M et al., "Improving the Gate Stability of ZnO Thin-Film Transistors With Aluminum Oxide Dielectric Layers,", J. Electrochem. Soc. (Journal of the Electrochemical Society), 2008, vol. 155, No. 12, pp. H1009-H1014.

Ueno.K et al., "Field-Effect Transistor on SrTiO3 With Sputtered Al2O3 Gate Insulator,", Appl. Phys. Lett. (Applied Physics Letters), Sep. 1, 2003, vol. 83, No. 9, pp. 1755-1757.

<sup>\*</sup> cited by examiner

FIG. 1A

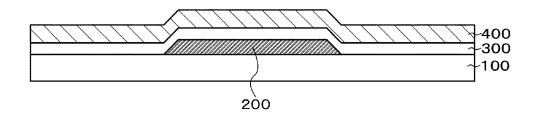


FIG. 1B

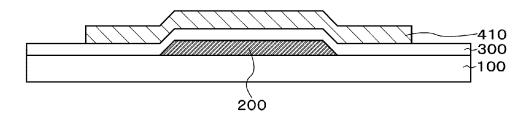


FIG. 1C

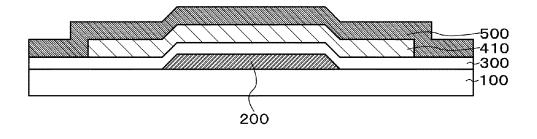


FIG. 2A

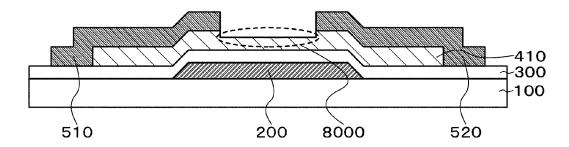
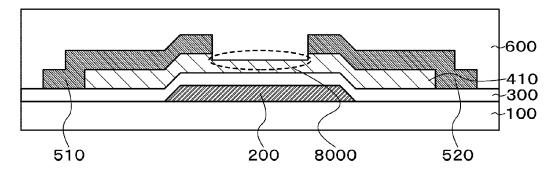
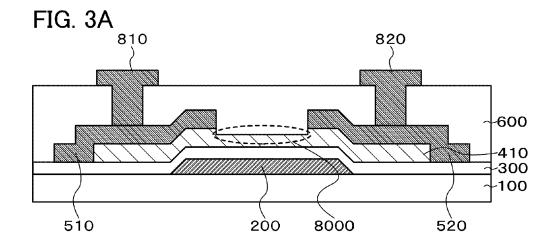


FIG. 2B





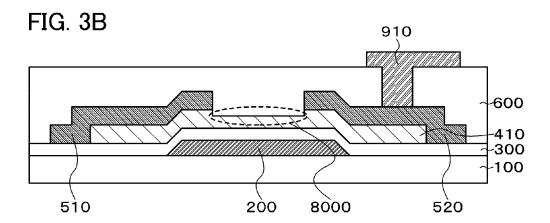


FIG. 4A

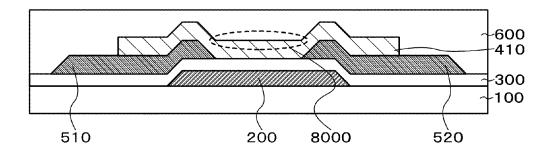


FIG. 4B

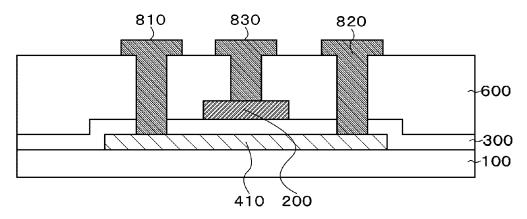


FIG. 4C

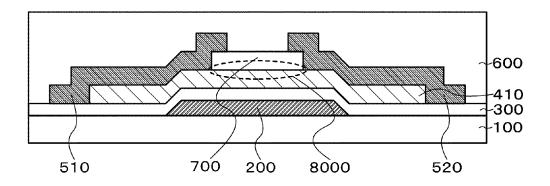


FIG. 5A

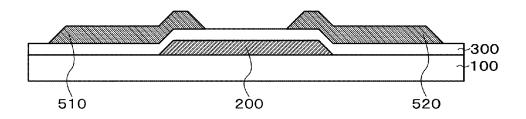


FIG. 5B

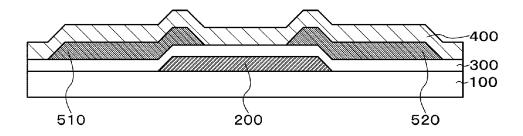


FIG. 5C

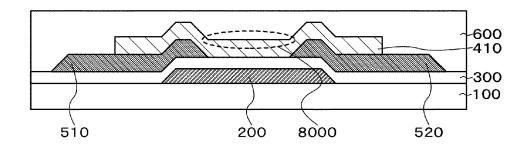
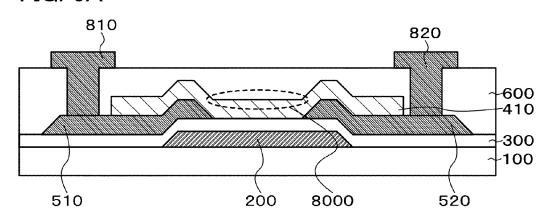


FIG. 6A



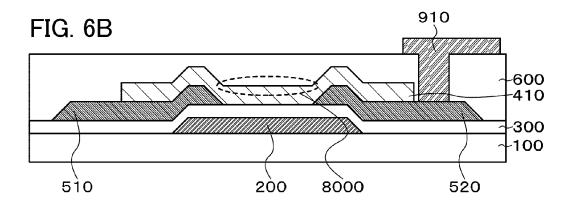


FIG. 7A

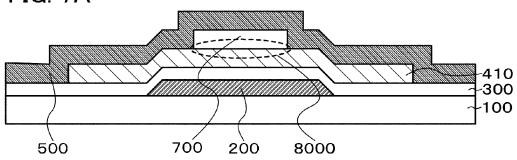


FIG. 7B

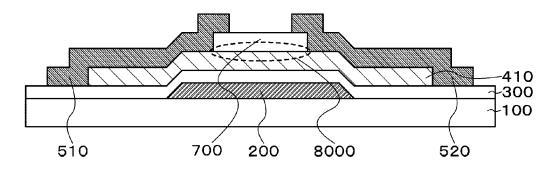


FIG. 7C

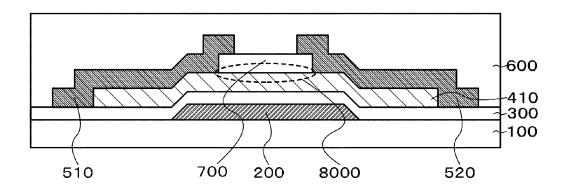
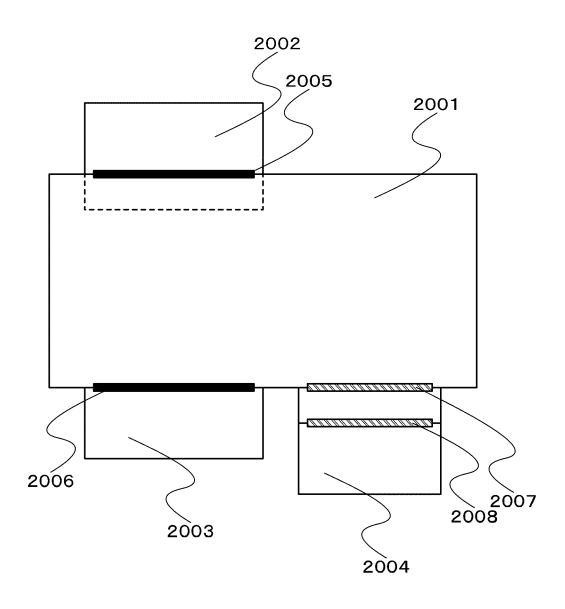
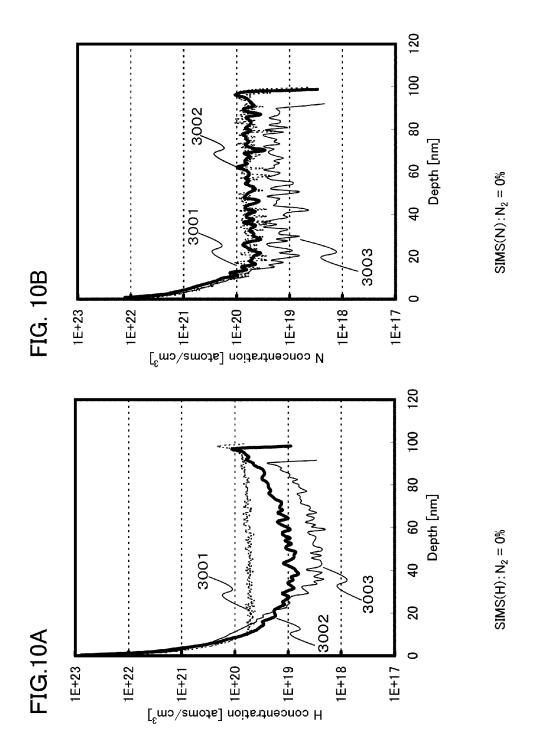


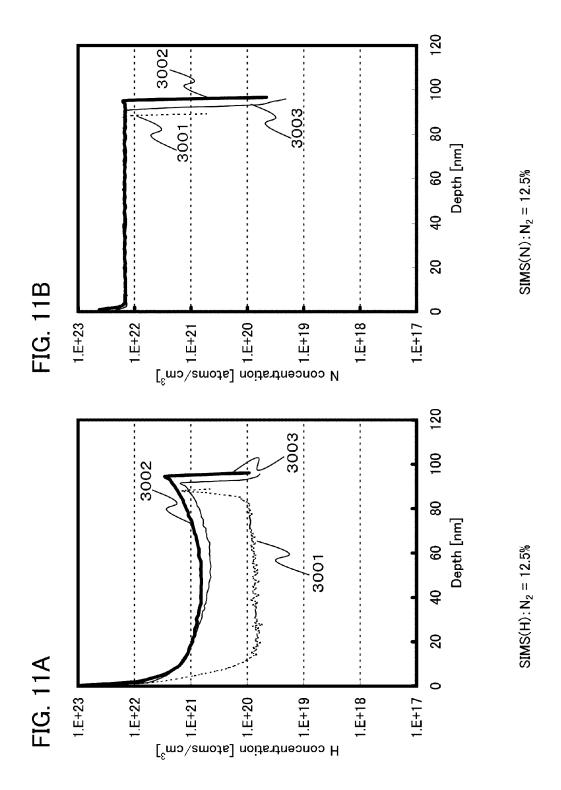
FIG. 8A 810 820 600 410 300 100 700 510 2Ò0 8000 5<u>2</u>0

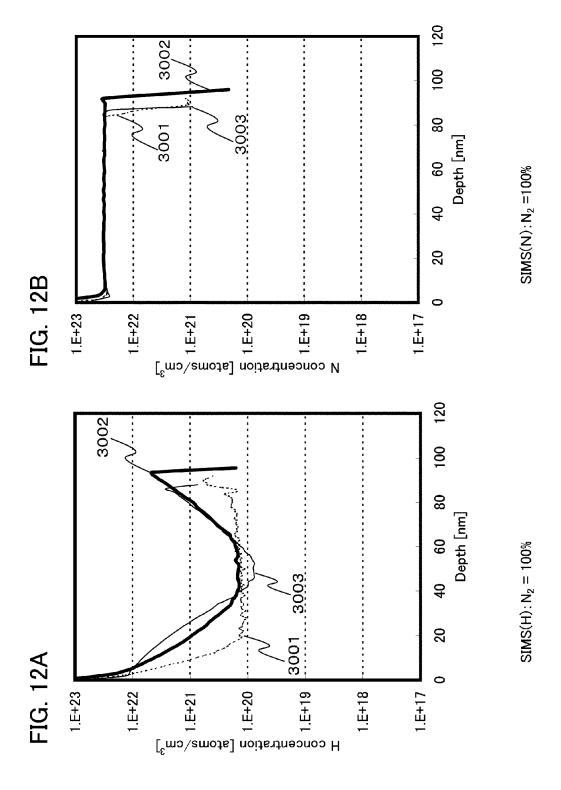
FIG. 8B 910 **√600** -410 -300 -100 510 700 8000 2Ò0 5<u>2</u>0

FIG. 9









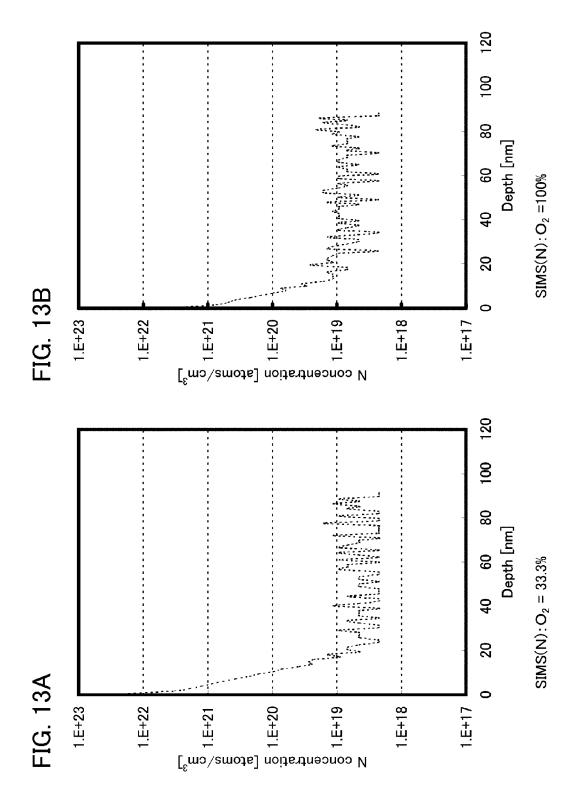
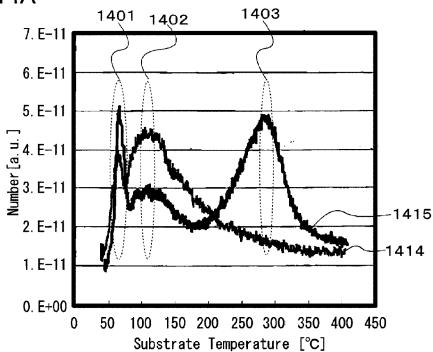


FIG. 14A





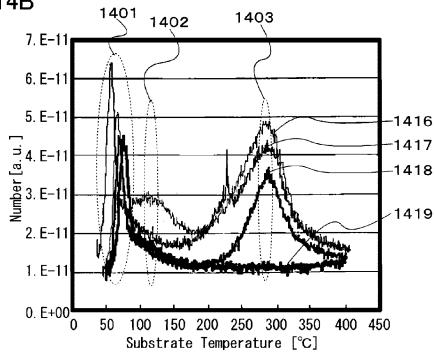


FIG. 15A

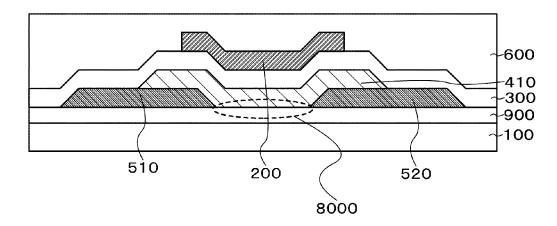
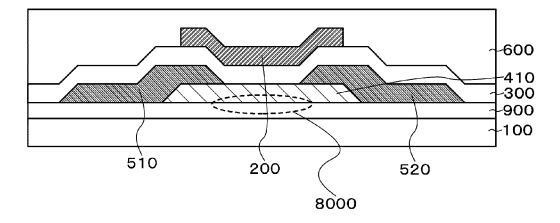
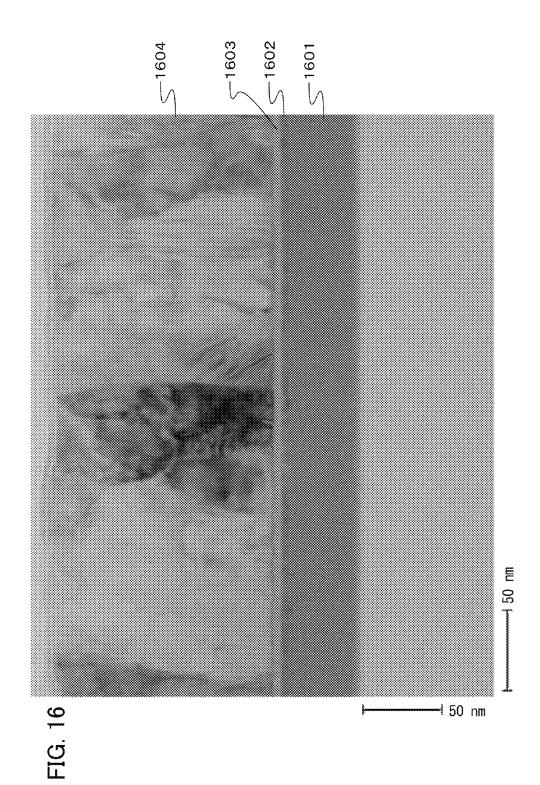


FIG. 15B





#### SEMICONDUCTOR DEVICE AND METHOD FOR MANUFACTURING THE SAME

#### TECHNICAL FIELD

The present invention relates to an oxide semiconductor.

#### BACKGROUND ART

A semiconductor device including an oxide semiconductor 10 is disclosed in Patent Document 1.

#### REFERENCE

#### Patent Document

[Patent Document 1] Japanese Published Patent Application No. 2007-123861

#### DISCLOSURE OF INVENTION

Many of the physical properties of a silicon semiconductor have already been understood, whereas many of the physical properties of an oxide semiconductor have been still unclear.

In particular, an adverse effect of an impurity on an oxide 25 semiconductor has been still unclear.

In view of the above, a structure will be disclosed in which an impurity that influences electrical characteristics of a semiconductor device including an oxide semiconductor layer is prevented or is eliminated.

There are two factors of carriers in an oxide semiconductor layer.

The first factor is oxygen deficiency in an oxide semicon-

ment in an oxide semiconductor layer.

Here, a hydrogen element serves as a carrier (a donor) in an oxide semiconductor layer.

Further, since a hydrogen element has a reducing character, it also serves as an element causing oxygen deficiency.

Therefore, it is said that a substance containing a hydrogen element is an element which prevents an oxide semiconductor layer from being highly purified so that the oxide semiconductor layer is not close to an i-type oxide semiconductor layer because a hydrogen element has two factors of inducing 45 carriers.

Note that as a substance containing a hydrogen element, for example, hydrogen, moisture, hydroxide, hydride, and the like can be given.

As a result of research, the present inventors found that, 50 surprisingly, hydrogen is more likely to enter an oxide semiconductor layer when a large amount of nitrogen is contained in the oxide semiconductor layer.

Conversely, hydrogen is less likely to enter an oxide semiconductor layer in which the nitrogen concentration is 55

Specifically, the nitrogen concentration in an oxide semiconductor layer measured by secondary ion mass spectrometry (SIMS) is set to  $1 \times 10^{20}$  atoms/cm<sup>3</sup> or less (or less than  $1 \times 10^{20}$  atoms/cm<sup>3</sup>), whereby an oxide semiconductor layer 60 which is difficult for hydrogen to enter can be formed.

That is, a semiconductor device which includes a gate electrode, an oxide semiconductor layer, and a gate insulating layer provided between the gate electrode and the oxide semiconductor layer and in which the nitrogen concentration in 65 facturing a semiconductor device. the oxide semiconductor layer is 1×10<sup>20</sup> atoms/cm<sup>3</sup> or less can be provided.

2

A semiconductor device which includes a gate electrode, a gate insulating layer provided over the gate electrode, an oxide semiconductor layer provided over the gate insulating layer, and a pair of contact electrodes provided over the oxide semiconductor layer and in which the nitrogen concentration in the oxide semiconductor layer is  $1 \times 10^{20}$  atoms/cm<sup>3</sup> or less can be provided.

A semiconductor device which includes a gate electrode, a gate insulating layer provided over the gate electrode, a pair of contact electrodes provided over the gate insulating layer, and an oxide semiconductor layer provided over the gate insulating layer and the pair of contact electrodes and in which the nitrogen concentration in the oxide semiconductor layer is  $1 \times 10^{20}$  atoms/cm<sup>3</sup> or less can be provided.

A semiconductor device which includes a gate electrode, a gate insulating layer provided over the gate electrode, an oxide semiconductor layer provided over the gate insulating layer, a channel protective layer provided over the oxide semiconductor layer, and a pair of contact electrodes pro-<sup>20</sup> vided over the oxide semiconductor layer and the channel protective layer and in which the nitrogen concentration in the oxide semiconductor layer is  $1 \times 10^{20}$  atoms/cm<sup>3</sup> or less can be

In addition, the semiconductor device in which the hydrogen concentration in the oxide semiconductor layer is  $6 \times 10^{18}$ atoms/cm<sup>3</sup> or less can be provided.

A method for manufacturing a semiconductor device including a gate electrode, an oxide semiconductor layer, and a gate insulating layer provided between the gate electrode and the oxide semiconductor layer, which includes the step of performing heat treatment on the oxide semiconductor layer having a nitrogen concentration of  $1 \times 10^{20}$  atoms/cm<sup>3</sup> or less at 350° C. or higher for 1 hour or longer, can be provided.

A method for manufacturing a semiconductor device The second factor is a donor element or an acceptor ele- 35 including a gate electrode, an oxide semiconductor layer, and a gate insulating layer provided between the gate electrode and the oxide semiconductor layer, which includes the step of performing heat treatment on the oxide semiconductor layer having a nitrogen concentration of 1×10<sup>20</sup> atoms/cm<sup>3</sup> or less at 450° C. or higher for 1 hour or longer, can be provided.

> A method for manufacturing a semiconductor device including a gate electrode, an oxide semiconductor layer, and a gate insulating layer provided between the gate electrode and the oxide semiconductor layer, which includes the step of performing heat treatment on the oxide semiconductor layer having a nitrogen concentration of 1×10<sup>20</sup> atoms/cm<sup>3</sup> or less at 550° C. or higher for 1 hour or longer, can be provided.

> A method for manufacturing a semiconductor device including a gate electrode, an oxide semiconductor layer, and a gate insulating layer provided between the gate electrode and the oxide semiconductor layer, which includes the step of performing heat treatment on the oxide semiconductor layer having a nitrogen concentration of  $1 \times 10^{20}$  atoms/cm<sup>3</sup> or less at 650° C. or higher for 3 minutes or longer, can be provided.

> The nitrogen concentration in an oxide semiconductor layer is reduced, whereby an oxide semiconductor layer which is difficult for hydrogen to enter can be formed.

> In other words, the nitrogen concentration in an oxide semiconductor layer is reduced, whereby hydrogen can be prevented from entering the oxide semiconductor layer.

#### BRIEF DESCRIPTION OF DRAWINGS

FIGS. 1A to 1C show an example of a method for manu-

FIGS. 2A and 2B show an example of a method for manufacturing a semiconductor device.

FIGS. 3A and 3B show an example of a method for manufacturing a semiconductor device.

FIGS. 4A to 4C each show an example of a semiconductor device.

FIGS. 5A to 5C show an example of a method for manu- 5 facturing a semiconductor device.

FIGS. **6A** and **6B** show an example of a method for manufacturing a semiconductor device.

FIGS. 7A to 7C show an example of a method for manufacturing a semiconductor device.

FIGS. 8A and 8B show an example of a method for manufacturing a semiconductor device.

FIG. 9 shows an example of a sputtering apparatus.

FIGS. 10A and 10B each show SIMS data.

FIGS. 11A and 11B each show SIMS data.

FIGS. 12A and 12B each show SIMS data.

FIGS. 13A and 13B each show SIMS data.

FIGS. 14A and 14B each show TDS data.

FIGS. 15A and 15B each show an example of a semiconductor device.

FIG. 16 shows a TEM photograph.

### BEST MODE FOR CARRYING OUT THE INVENTION

Embodiments and examples will be described in detail with reference to the accompanying drawings.

It is easily understood by those skilled in the art that modes and details of the present invention can be modified in various ways without departing from the spirit of the present invention.

Therefore, the present invention should not be interpreted as being limited to the description of the embodiments and examples below.

Note that in the structures described below, the same portions or portions having similar functions are denoted by the same reference numerals in different drawings, and explanation thereof will not be repeated.

The following embodiments and examples can be combined with each other, as appropriate.

#### Embodiment 1

An example of a method for manufacturing a semiconductor device will be described.

First, a gate electrode 200 is formed over a substrate 100 having an insulating surface. Then, a gate insulating layer 300 is formed over the gate electrode 200, and an oxide semiconductor layer 400 is formed over the gate insulating layer 300 (FIG. 1A).

As the substrate, any material can be used. For example, a glass substrate, a quartz substrate, a metal substrate, a plastic substrate, or a semiconductor substrate can be used, but the substrate is not limited to these examples.

In the case where an insulating substrate is used as the 55 substrate, the substrate has an insulating surface.

On the other hand, in the case where a metal substrate, a semiconductor substrate, or the like is used as the substrate, the substrate can have an insulating surface when a base insulating layer is formed over the substrate.

Note that a base insulating layer may be formed over the substrate also in the case where an insulating substrate is used as the substrate.

As the gate electrode, any material having conductivity can be used. For example, aluminum, titanium, molybdenum, 65 tungsten, gold, silver, copper, silicon, a variety of alloys, or an oxide conductive layer (typically, indium tin oxide and the 4

like) can be used, but the gate electrode is not limited to these examples. The gate electrode may have a single-layer structure or a stacked-layer structure.

As the gate insulating layer, any material having an insulating property can be used. For example, a silicon oxide film, a silicon nitride film, a silicon oxide film containing nitrogen, a silicon nitride film containing oxygen, an aluminum nitride film, an aluminum oxide film, a hafnium oxide film, or the like can be used, but the gate insulating layer is not limited to these examples. The gate insulating layer may have a single-layer structure or a stacked-layer structure.

Note that the amount of hydrogen and nitrogen in the gate insulating layer is preferably small so that injection of carriers into the oxide semiconductor layer can be prevented.

The gate insulating layer in which the amount of hydrogen is small is preferably formed using a deposition gas which does not contain hydrogen (H) or hydride (e.g., SiH<sub>4</sub>).

The gate insulating layer in which the amount of nitrogen is small is preferably formed using a deposition gas which does not contain nitrogen (N) or nitride (e.g., N<sub>2</sub>O or NH<sub>4</sub>).

Therefore, when attention is focused on the point that the amount of hydrogen is small, a gate insulating film formed by a sputtering method is preferably used because hydride (e.g.,  $SiH_4$ ) is used in a plasma CVD method.

Further, when attention is focused on the point that the amount of nitrogen is small, an oxide film which does not contain nitrogen is preferably used.

Note that a gate insulating layer formed by a plasma CVD method has fewer defects and has higher film quality than a gate insulating layer formed by a sputtering method.

Therefore, in some cases, transistor characteristics become favorable when a gate insulating layer formed by a plasma CVD method is used.

Thus, a plasma CVD method, a sputtering method, or another method may be used as appropriate as needed.

Note that in the case where a gate insulating layer formed by a plasma CVD method is used, a substance containing a hydrogen element is eliminated from the gate insulating layer when heat treatment is performed; therefore, in the case where a plasma CVD method is used, heat treatment (at higher than or equal to 200° C. and lower than or equal to 1000° C. (preferably, higher than or equal to 300° C. and lower than or equal to 800° C.)) is preferably performed after the gate insulating layer is formed.

Note that as a substance containing a hydrogen element, for example, hydrogen, moisture, hydroxide, hydride, and the like can be given.

Examples of the oxide semiconductor layer include, but not limited to, In-Ga-Zn-O-based oxide (containing indium, gallium, zinc, and oxygen as the main components), In—Sn—Zn—O-based oxide (containing indium, tin, zinc, and oxygen as the main components), In-Al-Zn-Obased oxide (containing indium, aluminum, zinc, and oxygen as the main components), Sn-Ga-Zn-O-based oxide (containing tin, gallium, zinc, and oxygen as the main components), Al—Ga—Zn—O-based oxide (containing aluminum, gallium, zinc, and oxygen as the main components), Sn—Al—Zn—O-based oxide (containing tin, aluminum, zinc, and oxygen as the main components), In-Zn-Obased oxide (containing indium, zinc, and oxygen as the main components), Sn—Zn—O-based oxide (containing tin, zinc, and oxygen as the main components), Al—Zn—O-based oxide (containing aluminum, zinc, and oxygen as the main components), In-O-based oxide (oxide of indium (indium oxide)), Sn-O-based oxide (oxide of tin (tin oxide)), Zn—O-based oxide (oxide of zinc (zinc oxide)), and the like.

The oxide semiconductor layer can be formed by a sputtering method, an evaporation method, or the like, for

The thickness of the oxide semiconductor layer is preferably 5 nm to 1 µm (more preferably, 20 nm to 80 nm).

In the case where the oxide semiconductor layer is formed, careful attention needs to be paid so that nitrogen is not contained in the oxide semiconductor layer.

Specifically, the oxide semiconductor layer is preferably formed so that the nitrogen concentration in the oxide semiconductor layer measured by secondary ion mass spectrometry (SIMS) is  $1 \times 10^{20}$  atoms/cm<sup>3</sup> or less (or less than  $1 \times 10^{20}$ atoms/cm<sup>3</sup>),  $5 \times 10^{19}$  atoms/cm<sup>3</sup> or less (or less than  $5 \times 10^{19}$ atoms/cm<sup>3</sup>),  $1 \times 10^{19}$  atoms/cm<sup>3</sup> or less (or less than  $1 \times 10^{19}$ atoms/cm<sup>3</sup>),  $5 \times 10^{18}$  atoms/cm<sup>3</sup> or less (or less than  $5 \times 10^{18}$  15 atoms/cm $^3$ ), or  $1\times10^{18}$  atoms/cm $^3$  or less (or less than  $1\times10^{18}$ 

Note that nitrogen is less likely to enter the oxide semiconductor layer even in the case where the oxide semiconductor layer is subjected to heat treatment.

Therefore, the nitrogen concentration in the oxide semiconductor layer after the semiconductor device is completed, which is measured by secondary ion mass spectrometry (SIMS), is also preferably  $1\times10^{20}$  atoms/cm³ or less (or less than  $1\times10^{20}$  atoms/cm³),  $5\times10^{19}$  atoms/cm³ or less (or less 25 than  $5\times10^{19}$  atoms/cm³),  $1\times10^{19}$  atoms/cm³ or less (or less than  $1\times10^{19}$  atoms/cm³),  $5\times10^{18}$  atoms/cm³ or less (or less than  $1\times10^{19}$  atoms/cm³),  $5\times10^{18}$  atoms/cm³ or less (or less than  $5\times10^{18}$  atoms/cm<sup>3</sup>), or  $1\times10^{18}$  atoms/cm<sup>3</sup> or less (or less than  $1\times10^{18}$  atoms/cm<sup>3</sup>).

Note that as the value of the nitrogen concentration, the 30 average value in the effective range of secondary ion mass spectrometry (SIMS) can be adopted.

Alternatively, as the value of the nitrogen concentration, the maximum value in the effective range of secondary ion mass spectrometry (SIMS) may be adopted (when the maxi- 35 mum value in the effective range is smaller than a predetermined value, the average value in the effective range is also smaller than the predetermined value).

Here, an example of a sputtering apparatus is shown in

The sputtering apparatus in FIG. 9 includes a deposition chamber 2001, a cover 2002, a target 2003, and a pump 2004.

Note that a structure is employed in which a substrate is put to be provided for the cover 2002 and the target 2003 is sputtered for deposition.

The deposition chamber 2001 and the cover 2002 are connected to each other, and an O-ring 2005 is provided for a connecting portion.

The deposition chamber 2001 and the target 2003 are connected to each other, and an O-ring 2006 is provided for a 50 connecting portion.

The deposition chamber 2001 and the pump 2004 are connected to each other, and a metal gasket 2007 and a metal gasket 2008 are provided for connecting portions.

leakage from the connecting portion. In other words, by the O-rings and the metal gaskets, air (in particular, nitrogen) is prevented from entering the deposition chamber 2001.

The O-ring is a ring-like packing. A material thereof is, for example, rubber.

The metal gasket is a ring-like fixing sealant. A material thereof is, for example, metal.

The cover 2002 and the target 2003 are frequently opened and closed; therefore, the O-ring is used because it can be easily put on and taken off.

On the other hand, the pump is hardly opened and closed. Although the metal gasket is not easily put on and taken off,

the metal gasket is provided for the pump because it can improve airtightness compared with the O-ring.

In the case where the O-ring or the metal gasket has a chip, a crack, or the like, leakage occurs, so that air outside the deposition chamber 2001 enters the deposition chamber 2001.

Since air contains a large amount of nitrogen, careful attention needs to be paid so that the O-ring or the metal gasket does not have a chip, a crack, or the like in order to prevent air (in particular, nitrogen) due to the chip, the crack, or the like of the O-ring or the metal gasket from entering the deposition chamber 2001.

Although the maintenance needs to be further done, it is effective to replace all the O-rings with metal gaskets which can improve airtightness because air (in particular, nitrogen) can be prevented from entering the deposition chamber 2001.

On the other hand, in the case where nitrogen is attached to or enters the inner wall of the deposition chamber 2001 or a surface of the target 2003 or in the case where nitrogen floats 20 in the deposition chamber 2001 even when air (in particular, nitrogen) is prevented from entering the deposition chamber 2001, nitrogen might enter an oxide semiconductor layer.

Thus, in order to remove nitrogen which is attached to or enters the inner wall of the deposition chamber, the deposition chamber is heated at higher than or equal to 200° C. and lower than or equal to 500° C.

The deposition chamber is heated, whereby nitrogen which is attached to or enters the inner wall of the deposition chamber is released into the deposition chamber.

Then, nitrogen which is released through heating of the deposition chamber is evacuated with the use of the pump 2004, whereby nitrogen which is attached to or enters the inner wall of the deposition chamber can be removed.

Further, after the heat treatment and the evacuation treatment are performed on the deposition chamber and before the oxide semiconductor layer used for a semiconductor device is formed, a step of forming an oxide semiconductor layer over a dummy substrate is preferably performed.

When the step of forming the oxide semiconductor layer 40 over the dummy substrate is performed, nitrogen which is attached to or enters the surface of the target can be removed.

Further, since nitrogen which remains in the deposition chamber due to the step of forming the oxide semiconductor layer over the dummy substrate is taken into the oxide semiconductor layer over the dummy substrate, nitrogen which remains in the deposition chamber can be removed before the oxide semiconductor layer used for the semiconductor device is formed.

After that, the dummy substrate is taken out of the deposition chamber, and the oxide semiconductor layer used for the semiconductor device is formed.

Note that it is effective to perform deposition onto the dummy substrate a plurality of times.

As described above, the reduction of leakage from the Both the O-rings and the metal gaskets are used to prevent 55 deposition chamber, the reduction of nitrogen on the inner wall of the deposition chamber, deposition onto the dummy substrate, or the like is sufficiently performed, so that incorporation of nitrogen into the oxide semiconductor layer can be thoroughly prevented.

> Next, the oxide semiconductor layer 400 is etched into an island shape by a photolithography method, whereby an oxide semiconductor layer 410 is formed (FIG. 1B).

> Next, the oxide semiconductor layer is subjected to first heat treatment (at higher than or equal to X° C. and lower than Y° C.).

> A nitrogen atmosphere, a rare gas atmosphere, an oxygen atmosphere, an atmosphere containing oxygen and nitrogen,

an atmosphere containing oxygen and a rare gas, an atmosphere containing nitrogen and a rare gas, an atmosphere containing oxygen, nitrogen, and a rare gas, or the like can be selected as appropriate for the atmosphere in which the first heat treatment is performed.

The first heat treatment may be performed before the oxide semiconductor layer 400 is etched into an island shape to form the oxide semiconductor layer 410.

On the other hand, through the step of etching the oxide semiconductor layer 400 into an island shape to form the 10 oxide semiconductor layer 410 by a photolithography method, the oxide semiconductor layer is immersed in moisture of a photoresist and a stripping solution.

Therefore, in order to remove moisture due to the photoresist and the stripping solution, it is more preferable that the 15 first heat treatment be performed after the oxide semiconductor layer 400 is etched into an island shape to form the oxide semiconductor layer 410.

The lower limit ( $X^{\circ}$  C.) of the temperature of the first heat treatment can be set to 350° C. or higher (or higher than 350° 20 C.), 400° C. or higher (or higher than 400° C.), 450° C. or higher (or higher than 450° C.), 500° C. or higher (or higher than 500° C.), 550° C. or higher (or higher than 550° C.), 600° C. or higher (or higher than 600° C.), 650° C. or higher (or higher than 650° C.), 700° C. or higher (or higher than 700° 25 C.), or 750° C. or higher (or higher than 750° C.).

The first heat treatment is preferably performed by a heating method with the use of a furnace, an oven, gas RTA, or the like.

Gas RTA refers to a method in which an object is put in a 30 gas heated at high temperature for a short time (several minutes to several tens of minutes) to be rapidly heated.

There is no particular limitation on the upper limit of the temperature of the first heat treatment because the temperature of the first heat treatment is preferably high.

However, the upper limit (Y° C.) of the temperature of the first heat treatment is preferably lower than the allowable temperature limit of the substrate.

Further, the upper limit (Y° C.) of the temperature of the first heat treatment can be set to 1000° C. or lower (or lower 40 the maximum value in the effective range of secondary ion than  $1000^{\circ}$  C.),  $900^{\circ}$  C. or lower (or lower than  $900^{\circ}$  C.),  $800^{\circ}$ C. or lower (or lower than 800° C.), or 700° C. or lower (or lower than 700° C.).

The heating time of the first heat treatment is preferably 1 hour or longer. The upper limit of the heating time is not 45 particularly limited, but can be set to 10 hours or shorter, 9 hours or shorter, or 8 hours or shorter in consideration of reduction in process time.

In the case where the first heat treatment is performed with the use of gas RTA, the heating time of the first heat treatment 50 is preferably 3 minutes or longer. The upper limit of the heating time is not particularly limited, but can be set to 1 hour or shorter, 50 minutes or shorter, or 40 minutes or shorter.

When the measurement is performed by thermal desorption spectroscopy (TDS), a sample on which baking has been 55 performed at 450° C. for 1 hour does not have a peak of moisture at around 300° C. A sample on which baking has been performed at 650° C. for 3 minutes with the use of gas RTA also does not have a peak of moisture at around 300° C. On the other hand, a sample on which baking has been per- 60 formed at 350° C. for 1 hour has a peak of moisture at around 300° C.

When the measurement is performed by secondary ion mass spectrometry (SIMS), the hydrogen concentration in a sample on which baking has been performed at 550° C. for 1 65 hour is lower than that in a sample on which baking has been performed at 450° C. for 1 hour by almost 1 digit.

In other words, the oxide semiconductor layer in which the nitrogen concentration is reduced is subjected to the first heat treatment under a predetermined condition, whereby substances containing a hydrogen element which adversely affect electrical characteristics of a transistor can be drastically reduced.

Note that the higher the energy which is applied to the oxide semiconductor is, the more substances containing a hydrogen element are likely to be eliminated; therefore, the heating temperature is preferably high and the heating time is preferably long.

The hydrogen concentration in the oxide semiconductor layer after the first heat treatment is preferably  $6 \times 10^{18}$  atoms/ cm<sup>3</sup> or less (or less than  $6\times10^{18}$  atoms/cm<sup>3</sup>),  $5\times10^{18}$  atoms/ cm<sup>3</sup> or less (or less than 5×10<sup>18</sup> atoms/cm<sup>3</sup>), 4×10<sup>18</sup> atoms/ cm<sup>3</sup> or less (or less than 4×10<sup>18</sup> atoms/cm<sup>3</sup>), 3×10<sup>18</sup> atoms/ cm<sup>3</sup> or less (or less than 3×10<sup>18</sup> atoms/cm<sup>3</sup>), 1×10<sup>16</sup> atoms/ cm<sup>3</sup> or less (or less than  $1\times10^{16}$  atoms/cm<sup>3</sup>),  $1\times10^{14}$  atoms/ cm<sup>3</sup> or less (or less than 1×10<sup>14</sup> atoms/cm<sup>3</sup>), or 1×10<sup>12</sup> atoms/ cm<sup>3</sup> or less (or less than  $1 \times 10^{12}$  atoms/cm<sup>3</sup>).

The oxide semiconductor layer in which the nitrogen concentration is reduced can prevent incorporation of hydrogen in steps after the first heat treatment.

Thus, the hydrogen concentration in the oxide semiconductor layer after the semiconductor device is completed is also preferably  $6 \times 10^{18}$  atoms/cm<sup>3</sup> or less (or less than  $6 \times 10^{18}$ atoms/cm<sup>3</sup>),  $5 \times 10^{18}$  atoms/cm<sup>3</sup> or less (or less than  $5 \times 10^{18}$ atoms/cm<sup>3</sup>),  $4\times10^{18}$  atoms/cm<sup>3</sup> or less (or less than  $4\times10^{18}$ atoms/cm<sup>3</sup>), 3×10<sup>18</sup> atoms/cm<sup>3</sup> or less (or less than 3×10<sup>18</sup> atoms/cm<sup>3</sup>),  $1 \times 10^{16}$  atoms/cm<sup>3</sup> or less (or less than  $1 \times 10^{16}$ atoms/cm<sup>3</sup>),  $1 \times 10^{14}$  atoms/cm<sup>3</sup> or less (or less than  $1 \times 10^{14}$ atoms/cm<sup>3</sup>), or  $1 \times 10^{12}$  atoms/cm<sup>3</sup> or less (or less than  $1 \times 10^{12}$ 35 atoms/cm<sup>3</sup>).

Note that as the value of the hydrogen concentration, the average value in the effective range of secondary ion mass spectrometry (SIMS) can be adopted.

Alternatively, as the value of the hydrogen concentration, mass spectrometry (SIMS) may be adopted (when the maximum value in the effective range is smaller than a predetermined value, the average value in the effective range is also smaller than the predetermined value).

Note that the smaller the quantity of substances containing a hydrogen element in the oxide semiconductor layer is, the more electrical characteristics of the transistor including the oxide semiconductor layer are improved.

Next, a conductive layer 500 is formed over the oxide semiconductor layer 410 (FIG. 1C).

As the conductive layer, any material having conductivity can be used. For example, aluminum, titanium, molybdenum, tungsten, yttrium, indium, gold, silver, copper, silicon, a variety of alloys containing any of these metals, an oxide conductive layer (typically, indium tin oxide), or the like can be used, but the conductive layer is not limited to these examples. The conductive layer may have a single-layer structure or a stacked-layer structure.

Note that the conductive layer in contact with the oxide semiconductor layer is formed using titanium, indium, yttrium, an indium-zinc alloy, an alloy containing gallium (e.g., gallium nitride), or the like, whereby the contact resistance between the oxide semiconductor layer and an electrode (a wiring) which is formed by etching of the conductive layer can be reduced.

The reason why the contact resistance can be reduced is that the electron affinity of titanium, indium, yttrium, an

indium-zinc alloy, an alloy containing gallium (e.g., gallium nitride), or the like is lower than that of the oxide semiconductor layer.

That is, in the case of a single layer, a metal (or an alloy, a compound) having lower electron affinity than the oxide semiconductor layer is preferable.

On the other hand, in the case of a stacked layer, a metal (or an alloy, a compound) having lower electron affinity than the oxide semiconductor layer is preferably disposed to be in contact with the oxide semiconductor layer.

Since titanium (Ti), indium (In), yttrium (Y), an indiumzinc alloy (an In—Zn alloy), an alloy containing gallium (Ga) (e.g., gallium nitride), or the like has high resistivity, a material having low resistivity such as aluminum (Al), gold (Au), silver (Ag), copper (Cu), or a variety of alloys containing any of these metals is preferably stacked over the conductive layer which is disposed to be in contact with the oxide semiconductor layer.

Specifically, the following examples can be given, but not 20 limited to, a structure in which Ti and Al are sequentially stacked, a structure in which Ti and an alloy containing Al are sequentially stacked, a structure in which Y and Al are sequentially stacked, a structure in which Y and an alloy containing Al are sequentially stacked, a structure in which Ti, an alloy containing Al, and Ti are sequentially stacked, a structure in which Ti, an alloy containing Al, and Mo are sequentially stacked, a structure in which In, Al, and Mo are sequentially stacked, a structure in which Y, Al, and Ti are sequentially stacked, a structure in which Mo, Al, and Ti are sequentially stacked, and a structure in which Ti, an alloy containing Al, Mo, and Ti are sequentially stacked.

Note that the alloy having low resistivity refers to an alloy containing aluminum, gold, silver, copper, or the like and another substance (e.g., Al—Si, Al—Ti, Al—Nd, Cu—Pb—Fe, or Cu—Ni).

Note that the oxide conductive layer can be formed using a material similar to the material of the oxide semiconductor layer.

There is no particular limitation on the oxide conductive layer as long as the resistivity of the oxide conductive layer is lower than that of the oxide semiconductor layer serving as a channel formation region.

Here, the oxide conductive layer is oxide which is intentionally made to contain a large quantity of substances containing a hydrogen element or a large quantity of oxygen deficiency. Substances containing a hydrogen element and oxygen deficiency induce carriers, so that the conductivity of the oxide can be raised.

The oxide semiconductor layer is oxide which is intentionally made not to contain substances containing a hydrogen element or oxygen deficiency.

That is, the quantity of substances containing a hydrogen element or oxygen deficiency in the oxide semiconductor 55 layer is controlled, whereby the resistivity can be controlled.

Note that in the case where the oxide conductive layer and the oxide semiconductor layer serving as a channel formation region are formed using different materials and the oxide conductive layer has lower resistivity than the oxide semiconductor layer serving as a channel formation region, the resistivity does not need to be controlled by control of the quantity of substances containing a hydrogen element or oxygen deficiency in the oxide semiconductor layer.

Next, the conductive layer **500** is etched to form a plurality 65 of electrodes or a plurality of wirings (a source electrode (a contact electrode), a drain electrode (a contact electrode), a

10

wiring, and the like) (FIG. 2A). Note that in FIG. 2A, a contact electrode 510, a contact electrode 520, and the like are shown

By the step of FIG. 2A, a transistor (a channel-etched transistor) is completed.

Note that a portion surrounded by a dashed line 8000 in FIGS. 2A and 2B is slightly etched at the time of etching the conductive layer 500.

The portion surrounded by the dashed line **8000** is called a back channel because it is disposed on the rear side of the channel formation region.

Next, an insulating layer 600 (a protective layer or an interlayer insulating film) is formed to cover the transistor (FIG. 2B).

As the insulating layer, any material having an insulating property can be used. For example, a silicon oxide film, a silicon nitride film, a silicon oxide film containing nitrogen, a silicon nitride film containing oxygen, an aluminum nitride film, an aluminum oxide film, a siloxane film, an acrylic film, a polyimide film, or the like can be used, but the insulating layer is not limited to these examples. The interlayer insulating film may have a single-layer structure or a stacked-layer structure.

Here, the kind of insulating layer is changed to compare electrical characteristics of transistors. It is found that a film which is formed without using a substance containing a hydrogen element as a sputtering gas is preferably used for the insulating layer in a portion in contact with the back channel (the portion surrounded by the dashed line **8000**).

When a substance containing a hydrogen element is contained in the back channel, the threshold voltage (Vth) of the transistor shifts in a negative direction.

Since a deposition gas containing a hydrogen element (typically, SiH<sub>4</sub> or the like) is used in a plasma CVD method, a substance containing a hydrogen element is added to the back channel when the insulating layer is formed by a plasma CVD method.

A siloxane film, an acrylic film, a polyimide film, or the like contains a large amount of moisture; therefore, a substance containing a hydrogen element is constantly supplied to the back channel if any of these films is employed.

That is, a film in which the quantity of substances containing a hydrogen element is small is preferably used for the insulating layer in contact with the back channel.

Note that the step of FIG. 2B corresponds to a step of forming the insulating layer 600 over the contact electrode 510, the contact electrode 520, and the back channel (the portion surrounded by the dashed line 8000).

A structure may be employed in which, after the step of FIG. 2B, contact holes are formed in the insulating layer 600 and a wiring 810, a wiring 820, and the like are formed over the insulating layer 600 (FIG. 3A).

Alternatively, a structure may be employed in which, after the step of FIG. 2B, a contact hole is formed in the insulating layer 600 and a pixel electrode 910 is formed over the insulating layer 600 (FIG. 3B).

After the wirings are formed over the insulating layer 600 as illustrated in FIG. 3A, an insulating layer, a wiring, a transistor, a display element, an antenna, or the like may be further formed over the wirings.

After the pixel electrode 910 is formed as illustrated in FIG. 3B, a display element (e.g., an EL element or a liquid crystal element) is formed, whereby a display device can be formed.

After the step of FIG.  $3\mathrm{A}$  or FIG.  $3\mathrm{B}$ , second heat treatment is preferably performed.

The second heat treatment may be performed between the step of FIG. **2**B and the step of FIG. **3**A or FIG. **3**B.

In other words, the timing of the second heat treatment is not particularly limited as long as it is performed after the insulating layer 600 is formed.

The second heat treatment is preferably performed at higher than or equal to  $150^{\circ}$  C. and lower than or equal to  $500^{\circ}$  C. (preferably higher than or equal to  $200^{\circ}$  C. and lower than or equal to  $300^{\circ}$  C.).

The heating time of the second heat treatment is preferably longer than or equal to 1 hour and shorter than or equal to 10 hours

The second heat treatment is preferably performed using a furnace, an oven, gas RTA, or the like.

Note that hydrogen in the oxide semiconductor layer is released and oxygen in the oxide semiconductor layer is also released by the first heat treatment which has been performed previously.

That is, by the first heat treatment, oxygen deficiency is 20 formed in the oxide semiconductor layer.

Thus, when the second heat treatment is performed, the insulating layer is made to be in an oxygen-excess state; therefore, oxygen can be supplied to the oxide semiconductor layer and oxygen deficiency in the oxide semiconductor layer 25 can be reduced.

Examples of the formation method of the insulating layer containing excessive oxygen include, but are not limited to, a method in which, in the case where a non-oxide target (silicon, aluminum, or the like) is used as a sputtering target and oxygen is used as a sputtering gas for reactive sputtering, the flow rate of oxygen is increased; a method in which an oxide target (silicon oxide, aluminum oxide, or the like) is used as a sputtering target and oxygen is used as a sputtering gas (in the case where an oxide target is used, oxygen is not used as a sputtering gas in general); and a method in which an insulating layer is formed and oxygen is introduced into the insulating layer by ion implantation or ion doping (note that in the case where reactive sputtering is performed, a gas such as argon is not preferably used so that a sputtering gas contains oxygen at 100%).

That is, a method in which oxygen is used as a deposition gas for formation of the insulating layer, a method in which oxygen is added to the insulating layer after the insulating 45 layer is formed, or the like may be used. Needless to say, oxygen may be used as a deposition gas for formation of the insulating layer, and in addition, oxygen may be added to the insulating layer after the insulating layer is formed.

Note that in the case where titanium is used for the contact 50 electrode, the second heat treatment is performed after the insulating layer **600** is formed, so that titanium oxide can be formed between the oxide semiconductor layer and titanium.

By formation of titanium oxide, the contact resistance between the oxide semiconductor layer and titanium can be 55 reduced.

Note that when titanium oxide is present between the oxide semiconductor layer and titanium, the contact resistance between the oxide semiconductor layer and titanium can be reduced, so that a structure may be formed in which titanium oxide and titanium are sequentially stacked when the contact electrode is formed.

In this case, titanium oxide can be formed by a sputtering method, an evaporation method, or the like.

The contents of this embodiment or part thereof can be 65 implemented in combination with any of the other embodiments and examples.

12

#### Embodiment 2

In this embodiment, a semiconductor device including a transistor which has a structure different from that of the transistor in Embodiment 1 will be described.

Note that materials or the like of the layers are the same as those in Embodiment 1.

A transistor in FIG. 4A is a bottom-gate bottom-contact (BGBC) transistor, and includes the gate electrode 200 provided over the substrate 100 having an insulating surface; the gate insulating layer 300 provided over the gate electrode 200; the contact electrode 510 and the contact electrode 520 provided over the gate insulating layer 300; and the oxide semiconductor layer 410 (island shape) provided over the gate insulating layer 300, the contact electrode 510, and the contact electrode 520.

Note that the insulating layer 600 is provided to cover the transistor.

A portion surrounded by the dashed line **8000** serves as a back channel.

A transistor in FIG. 4B is a top-gate transistor, and includes the oxide semiconductor layer 410 (island shape) provided over the substrate 100 having an insulating surface; the gate insulating layer 300 provided over the oxide semiconductor layer 410; and the gate electrode 200 provided over the gate insulating layer 300.

Note that the insulating layer 600 is provided to cover the transistor, and the wiring 810, the wiring 820, and a wiring 830 are provided through contact holes provided in the insulating layer 600.

A transistor in FIG. 4C is a channel-stop transistor, and includes the gate electrode 200 provided over the substrate 100 having an insulating surface; the gate insulating layer 300 provided over the gate electrode 200; the oxide semiconductor layer 410 (island shape) provided over the gate insulating layer 300; a channel protective layer 700 provided over the oxide semiconductor layer 410; and the contact electrode 510 and the contact electrode 520 provided over the oxide semiconductor layer 410 and the channel protective layer 700.

Note that the insulating layer 600 is provided to cover the transistor.

A portion surrounded by the dashed line 8000 serves as a back channel.

Here, the channel protective layer  $700\,\mathrm{can}$  be formed using a material similar to the material of the insulating layer  $600\,\mathrm{described}$  in Embodiment 1. The channel protective layer  $700\,\mathrm{and}$  the insulating layer  $600\,\mathrm{may}$  be formed using either the same material or different materials.

In the channel-stop transistor, a portion in contact with the back channel is not the insulating layer 600 but the channel protective layer 700.

That is, a film in which the quantity of substances containing a hydrogen element is small is preferably used for the channel protective layer 700.

A transistor in FIG. 15A is a top-gate bottom-contact (TGBC) transistor, and includes the contact electrode 510 and the contact electrode 520 provided over a base insulating layer 900; the oxide semiconductor layer 410 (island shape) provided over the base insulating layer 900, the contact electrode 510, and the contact electrode 520; the gate insulating layer 300 provided over the oxide semiconductor layer 410; and the gate electrode 200 provided over the gate insulating layer 300.

Note that the base insulating layer 900 is provided over the substrate 100.

The insulating layer 600 is provided to cover the transistor.

A portion surrounded by the dashed line 8000 serves as a back channel.

Here, the base insulating layer 900 can be formed using a material similar to the material of the insulating layer 600 described in Embodiment 1. The base insulating layer 900 5 and the insulating layer 600 may be formed using either the same material or different materials.

In the top-gate bottom-contact transistor, a portion in contact with the back channel is not the insulating layer 600 but the base insulating layer 900.

That is, a film in which the quantity of substances containing a hydrogen element is small is preferably used for the base insulating layer 900.

The base insulating layer 900 is preferably formed using an insulating layer containing excessive oxygen by a method 15 similar to the method described in Embodiment 1. In this case, when first heat treatment is performed, oxygen is released from the oxide semiconductor layer and oxygen is supplied to the oxide semiconductor layer from the base insulating layer at the same time.

A transistor in FIG. 15B is a top-gate top-contact (TGTC) transistor, and includes the oxide semiconductor layer 410 (island shape) provided over the base insulating layer 900; the contact electrode 510 and the contact electrode 520 provided over the oxide semiconductor layer 410 and the base insulat- 25 ing layer 900; the gate insulating layer 300 provided over the oxide semiconductor layer 410, the contact electrode 510, and the contact electrode 520; and the gate electrode 200 provided over the gate insulating layer 300.

Note that the base insulating layer **900** is provided over the 30 substrate 100.

The insulating layer 600 is provided to cover the transistor. A portion surrounded by the dashed line 8000 serves as a back channel.

Here, the base insulating layer 900 can be formed using a 35 material similar to the material of the insulating layer 600 described in Embodiment 1. The base insulating layer 900 and the insulating layer 600 may be formed using either the same material or different materials.

with the back channel is not the insulating layer 600 but the base insulating layer 900.

That is, a film in which the quantity of substances containing a hydrogen element is small is preferably used for the base insulating layer 900.

Note that in the transistor in FIG. 4B, an off-set region having a width of several micrometers is formed between a channel formation region (a region where the gate electrode and the oxide semiconductor layer overlap with each other) and each of contact regions (regions where the wirings and 50 the oxide semiconductor layer are in contact with each other).

The off-set region has an advantage of reducing off current of the transistor, but it has a disadvantage of reducing also on current of the transistor.

On the other hand, unlike in FIG. 4B, an off-set region is 55 not present in FIGS. 15A and 15B; thus, unlike the transistor in FIG. 4B, the transistors in FIGS. 15A and 15B have an advantage of improving on current.

As described above, the transistor may employ any strucfure.

In other words, the transistor may employ any structure as long as the transistor includes at least the gate electrode, the oxide semiconductor layer, and the gate insulating layer provided between the gate electrode and the oxide semiconduc-

A dual-gate transistor may be employed in which a first gate electrode, a first gate insulating layer over the first gate 14

electrode, an oxide semiconductor layer over the first gate insulating layer, a second gate insulating layer over the oxide semiconductor layer, and a second gate electrode over the second gate insulating layer are provided.

Therefore, the structure of the transistor is not limited to any of the structures described in Embodiment 1 and 2.

The contents of this embodiment or part thereof can be implemented in combination with any of the other embodiments and examples.

#### Embodiment 3

An example of a method for manufacturing the semiconductor device in FIG. 4A will be described.

Note that materials, conditions of heat treatment, and the like are the same as those in another embodiment.

The gate electrode 200 is formed over the substrate 100 having an insulating surface. Then, the gate insulating layer 300 is formed over the gate electrode 200, and the contact 20 electrode 510 and the contact electrode 520 are formed over the gate insulating layer 300 (FIG. 5A).

Next, the oxide semiconductor layer 400 is formed over the gate insulating layer 300, the contact electrode 510, and the contact electrode 520 (FIG. 5B).

Next, the oxide semiconductor layer is subjected to first heat treatment.

The conditions of the first heat treatment are the same as those in Embodiment 1.

The first heat treatment may be performed after the oxide semiconductor layer 400 is etched into an island shape to form the oxide semiconductor layer 410.

However, after the oxide semiconductor layer 400 is etched into an island shape to form the oxide semiconductor layer 410, the contact electrodes are exposed.

When the first heat treatment is performed in the state where the contact electrodes are exposed, surfaces of the contact electrodes are oxidized and the conductivity of the surfaces are reduced.

Therefore, the first heat treatment is preferably performed In the top-gate top-contact transistor, a portion in contact 40 in the state where the contact electrodes are covered with the oxide semiconductor layer 400.

> Next, the oxide semiconductor layer 400 is etched into an island shape to form the oxide semiconductor layer 410, and the insulating layer 600 is formed to cover a transistor (FIG. 5C)

> Note that a portion surrounded by the dashed line 8000 serves as a back channel.

> A structure may be employed in which, after the step of FIG. 5C, contact holes are formed in the insulating layer 600 and the wiring 810, the wiring 820, and the like are formed over the insulating layer 600 (FIG. 6A).

> Alternatively, a structure may be employed in which, after the step of FIG. 5C, a contact hole is formed in the insulating layer 600 and a pixel electrode 910 is formed over the insulating layer 600 (FIG. 6B).

> After the wirings are formed over the insulating layer 600 as illustrated in FIG. 6A, an insulating layer, a wiring, a transistor, a display element, an antenna, or the like may be further formed over the wirings.

After the pixel electrode 910 is formed as illustrated in FIG. 6B, a display element (e.g., an EL element or a liquid crystal element) is formed, whereby a display device can be formed.

After the step of FIG. 6A or FIG. 6B, second heat treatment 65 is preferably performed.

The second heat treatment may be performed between the step of FIG. 5C and the step of FIG. 6A or FIG. 6B.

The conditions of the second heat treatment are the same as those in Embodiment 1.

The contents of this embodiment or part thereof can be implemented in combination with any of the other embodiments and examples.

#### **Embodiment 4**

An example of a method for manufacturing the semiconductor device in FIG. 4C will be described.

Note that materials, conditions of heat treatment, and the like are the same as those in another embodiment.

First, the structure of FIG. 1B is formed as in Embodiment 1.

Note that first heat treatment is also performed as in Embodiment 1.

Next, the channel protective layer 700 (island shape) is formed, and the conductive layer 500 is formed to cover the channel protective layer 700 (FIG. 7A).

Then, the conductive layer 500 is etched to form the contact electrode 510 and the contact electrode 520 (FIG. 7B).

Note that a portion surrounded by the dashed line 8000 serves as a back channel.

Because of the presence of the channel protective layer 25 **700**, the back channel is not etched during the formation of the contact electrodes; therefore, damage to the back channel can be reduced.

Here, the channel protective layer 700 can be formed using a material similar to the material of the insulating layer 600 described in Embodiment 1. The channel protective layer 700 and the insulating layer 600 may be formed using either the same material or different materials.

In the channel-stop transistor, a portion in contact with the back channel is not the insulating layer 600 but the channel protective layer 700.

That is, a film in which the quantity of substances containing a hydrogen element is small is preferably used for the channel protective layer 700.

The channel protective layer 700 is preferably formed using an insulating layer containing excessive oxygen by a method similar to the method described in Embodiment 1.

Next, the insulating layer 600 is formed to cover the transistor (FIG. 7C).

A structure may be employed in which, after the step of FIG. 7C, contact holes are formed in the insulating layer 600 and the wiring 810, the wiring 820, and the like are formed over the insulating layer 600 (FIG. 8A).

Alternatively, a structure may be employed in which, after 50 the step of FIG. 7C, a contact hole is formed in the insulating layer 600 and the pixel electrode 910 is formed over the insulating layer 600 (FIG. 8B).

After the wirings are formed over the insulating layer 600 as illustrated in FIG. 8A, an insulating layer, a wiring, a 55 transistor, a display element, an antenna, or the like may be further formed over the wirings.

After the pixel electrode **910** is formed as illustrated in FIG. **8**B, a display element (e.g., an EL element or a liquid crystal element) is formed, whereby a display device can be 60 formed.

After the step of FIG. 8A or FIG. 8B, second heat treatment is preferably performed.

The second heat treatment may be performed between the step of FIG. 7C and the step of FIG. 8A or FIG. 8B.

The conditions of the second heat treatment are the same as those in Embodiment 1.

16

The contents of this embodiment or part thereof can be implemented in combination with any of the other embodiments and examples.

#### Embodiment 5

As a semiconductor device, there are various integrated circuits.

For example, display devices (such as liquid crystal display devices and electroluminescent display devices (light-emitting devices)), semiconductor devices for performing wireless communication through antennas (such as RFID tags, wireless tags, IC chips, wireless chips, noncontact signal processing devices, and semiconductor integrated circuit chips), and the like are given, but the integrated circuit is not limited to these examples.

The contents of this embodiment or part thereof can be implemented in combination with any of the other embodiments and examples.

#### Example 1

An influence of the nitrogen concentration in an oxide semiconductor was examined.

First, an oxide semiconductor layer was formed over a glass substrate.

Then, the case where heat treatment was not performed after the formation of the oxide semiconductor layer and the case where heat treatment was performed after the formation of the oxide semiconductor layer were compared.

They were compared by secondary ion mass spectrometry (SIMS).

Note that the heat treatment was performed at 350  $^{\rm o}$  C. for 1 hour.

The heating atmosphere was set to an air atmosphere or a nitrogen atmosphere.

Here, the oxide semiconductor layer was formed by sputtering using an oxide semiconductor target in which the atom ratio of In to Ga and Zn was 1:1:0.5 (In<sub>2</sub>O<sub>3</sub>:Ga<sub>2</sub>O<sub>3</sub>:ZnO=1: 1:1 (at a molar ratio)).

Note that before the formation of the oxide semiconductor layer, the reduction of leakage from a deposition chamber, the reduction of nitrogen on the inner wall of the deposition chamber, deposition onto a dummy substrate, or the like was sufficiently performed, so that incorporation of nitrogen into the oxide semiconductor layer was thoroughly prevented. (Sample 1)

In Sample 1, the flow rate of a sputtering gas was set as follows:  $Ar/N_2=40/0$  sccm (the proportion of  $N_2$  was 0%). Note that three Samples 1 were prepared.

The measurement results of SIMS of Sample 1 are shown in FIGS. **10**A and **10**B. (Sample 2)

In Sample 2, the flow rate of a sputtering gas was set as follows:  $Ar/N_2=35/5$  sccm (the proportion of  $N_2$  was 12.5%). Note that three Samples 2 were prepared.

The measurement results of SIMS of Sample 2 are shown in FIGS. 11A and 11B. (Sample 3)

In Sample 3, the flow rate of a sputtering gas was set as follows:  $Ar/N_2=0/40$  sccm (the proportion of  $N_2$  was 100%). Note that three Samples 3 were prepared.

The measurement results of SIMS of Sample 3 are shown in FIGS. 12A and 12B.

(Consideration)

In each of FIGS. 10A and 10B, FIGS. 11A and 11B, and FIGS. 12A and 12B, a dotted line 3001 denotes the sample on

0

which heat treatment was not performed (such a sample is referred to as an "as-deposited sample"), a thick solid line 3002 denotes the sample which was heated in a nitrogen atmosphere (such a sample is referred to as an " $N_2$ -baked sample"), and a thin solid line 3003 denotes the sample which was heated in an air atmosphere (such a sample is referred to as an "air-baked sample").

FIG. 10A, FIG. 11A, and FIG. 12A show the hydrogen concentration, and FIG. 10B, FIG. 11B, and FIG. 12B show the nitrogen concentration.

In FIGS. 10A and 10B, FIGS. 11A and 11B, and FIGS. 12A and 12B, the vertical axis represents the concentration, and the horizontal axis represents the depth from a surface of the oxide semiconductor layer (the film thickness).

Note that the measurement results of secondary ion mass 15 spectrometry (SIMS) have an effective range.

In the case of this example, in the vicinity of the surface of the oxide semiconductor layer (at a depth of approximately 0 nm to 30 nm in FIGS. 10A and 10B, FIGS. 11A and 11B, and FIGS. 12A and 12B) and in the vicinity of the interface 20 between the oxide semiconductor layer and the glass substrate (at a depth of approximately 80 nm to 100 nm in FIGS. 10A and 10B, FIGS. 11A and 11B, and FIGS. 12A and 12B), accurate values are difficult to evaluate.

Therefore, in the case of this example, the effective range  $\,^{25}$  of the measurement results of SIMS was set to the range of 30 nm to 80 nm in depth.

Here, the average values of the hydrogen concentration in the effective range of the dotted line 3001 (the as-deposited sample) in FIG. 10A, the dotted line 3001 (the as-deposited sample) in FIG. 11A, and the dotted line 3001 (the as-deposited sample) in FIG. 12A are compared. It is found that the average value in FIG. 11A is larger than that in FIG. 10A, and the average value in FIG. 12A is larger than that in FIG. 11A.

Further, the average values of the nitrogen concentrations 35 in the effective range of the dotted line 3001 (the as-deposited sample) in FIG. 10B, the dotted line 3001 (the as-deposited sample) in FIG. 11B, and the dotted line 3001 (the as-deposited sample) in FIG. 12B are compared. It is found that the average value in FIG. 11B is larger than that in FIG. 10B, and 40 the average value in FIG. 12B is larger than that in FIG. 11B.

That is, in the as-deposited samples, the higher the nitrogen concentration in the oxide semiconductor layer is, the higher the hydrogen concentration in the oxide semiconductor layer becomes.

Therefore, the higher the nitrogen concentration in the oxide semiconductor layer is, the more hydrogen is likely to enter the oxide semiconductor layer.

With reference to FIG. 10A, the hydrogen concentration is decreased through the heat treatment.

On the other hand, with reference to FIG. 11A and FIG. 12A, the hydrogen concentration is increased through the heat treatment.

Note that in FIG. 12A, in the range of 30 nm to 60 nm, there is little difference in the hydrogen concentration among the 55 dotted line 3001 (the as-deposited sample), the thick solid line 3002 (the  $\rm N_2$ -baked sample), and the thin solid line 3003 (the air-baked sample).

However, in the range of 30 nm to 80 nm which is the effective range, the average value of the hydrogen concentration of the thick solid line 3002 (the  $N_2$ -baked sample) and that of the thin solid line 3003 (the air-baked sample) are larger than that of the dotted line 3001 (the as-deposited sample).

Thus, it is clear that, in the case where the total amounts of 65 hydrogen in the oxide semiconductor layers are compared, the amount of hydrogen shown by the thick solid line 3002

18

(the  $N_2$ -baked sample) and that shown by the thin solid line 3003 (the air-baked sample) are larger than that shown by the dotted line 3001 (the as-deposited sample).

It seem that the reason why the thick solid line 3002 (the  $N_2$ -baked sample) and the thin solid line 3003 (the air-baked sample) have V-like shapes in FIG. 12A is that hydrogen is injected from the surface of the film and the glass substrate.

Here, in Sample 1, the maximum value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $1\times10^{20}$  atoms/cm<sup>3</sup> or less (less than  $1\times10^{20}$  atoms/cm<sup>3</sup>) (FIG. **10**B).

Therefore, in Sample 1, the average value of the nitrogen concentration in the effective range of the measurement results of SIMS was also  $1\times10^{20}$  atoms/cm<sup>3</sup> or less (less than  $1\times10^{20}$  atoms/cm<sup>3</sup>) (the average value is not larger than the maximum value).

Note that, as for the as-deposited Sample 1 (FIGS. **10**A and **10**B), the maximum value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $9.3 \times 10^{19}$  atoms/cm³, the minimum value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $1.9 \times 10^{19}$  atoms/cm³, and the average value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $6.1 \times 10^{19}$  atoms/cm³.

Further, as for the as-deposited Sample 1 (FIGS. **10**A and **10**B), the maximum value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $6.9 \times 10^{19}$  atoms/cm<sup>3</sup>, the minimum value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $4.5 \times 10^{19}$  atoms/cm<sup>3</sup>, and the average value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $5.6 \times 10^{19}$  atoms/cm<sup>3</sup>.

Note that, as for the  $N_2$ -baked Sample 1 (FIGS. 10A and 10B), the maximum value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $9.7 \times 10^{19}$  atoms/cm³, the minimum value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $3.0 \times 10^{19}$  atoms/cm³, and the average value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $6.0 \times 10^{19}$  atoms/cm³.

Further, as for the  $\rm N_2$ -baked Sample 1 (FIGS. 10A and 10B), the maximum value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $2.3\times10^{19}$  atoms/cm³, the minimum value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $6.4\times10^{18}$  atoms/cm³, and the average value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $1.2\times10^{19}$  atoms/cm³.

Note that, as for the air-baked Sample 1 (FIGS. **10**A and **10**B), the maximum value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $3.1 \times 10^{19}$  atoms/cm<sup>3</sup>, the minimum value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $4.4 \times 10^{18}$  atoms/cm<sup>3</sup>, and the average value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $1.8 \times 10^{19}$  atoms/cm<sup>3</sup>.

Further, as for the air-baked Sample 1 (FIGS. **10**A and **10**B), the maximum value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $6.7 \times 10^{18}$  atoms/cm<sup>3</sup>, the minimum value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $2.0 \times 10^{18}$  atoms/cm<sup>3</sup>, and the average value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $3.8 \times 10^{18}$  atoms/cm<sup>3</sup>.

On the other hand, in Sample 2 and Sample 3, the minimum value of the nitrogen concentration in the effective range of

the measurement results of SIMS was  $1\times10^{22}$  atoms/cm<sup>3</sup> or more (FIG. 11B and FIG. 12B).

Therefore, in Sample 2 and Sample 3, the average value of the nitrogen concentration in the effective range of the measurement results of SIMS was also  $1\times10^{22}$  atoms/cm<sup>3</sup> or more 5 (the average value is not smaller than the maximum value).

Note that, as for the as-deposited Sample 2 (FIGS. 11A and 11B), the maximum value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $1.6 \times 10^{22}$  atoms/cm<sup>3</sup>, the minimum value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $1.5 \times 10^{22}$  atoms/cm<sup>3</sup>, and the average value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $1.5 \times 10^{22}$  atoms/cm<sup>3</sup>.

Further, as for the as-deposited Sample 2 (FIGS. 11A and 15 11B), the maximum value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $1.0\times10^{20}$  atoms/cm<sup>3</sup>, the minimum value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $6.3\times10^{19}$  atoms/cm<sup>3</sup>, and the average 20 value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $7.8\times10^{19}$  atoms/cm<sup>3</sup>.

Note that, as for the  $N_2$ -baked Sample 2 (FIGS. 11A and 11B), the maximum value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $1.5\times25$   $10^{22}$  atoms/cm³, the minimum value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $1.4\times10^{22}$  atoms/cm³, and the average value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $1.5\times10^{22}$  atoms/cm³.

Further, as for the  $N_2$ -baked Sample 2 (FIGS. 11A and 11B), the maximum value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $1.2 \times 10^{21}$  atoms/cm<sup>3</sup>, the minimum value of the hydrogen concentration in the effective range of the measurement 35 results of SIMS was  $6.5 \times 10^{20}$  atoms/cm<sup>3</sup>, and the average value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $7.7 \times 10^{20}$  atoms/cm<sup>3</sup>.

Note that, as for the air-baked Sample 2 (FIGS. 11A and 11B), the maximum value of the nitrogen concentration in the 40 effective range of the measurement results of SIMS was  $1.6 \times 10^{22}$  atoms/cm<sup>3</sup>, the minimum value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $1.5 \times 10^{22}$  atoms/cm<sup>3</sup>, and the average value of the nitrogen concentration in the effective range of the measure-45 ment results of SIMS was  $1.6 \times 10^{22}$  atoms/cm<sup>3</sup>.

Further, as for the air-baked Sample 2 (FIGS. 11A and 11B), the maximum value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $6.9\times10^{20}$  atoms/cm<sup>3</sup>, the minimum value of the hydrogen 50 concentration in the effective range of the measurement results of SIMS was  $4.4\times10^{20}$  atoms/cm<sup>3</sup>, and the average value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $5.3\times10^{20}$  atoms/cm<sup>3</sup>.

Note that, as for the as-deposited Sample 3 (FIGS. 12A and 51 12B), the maximum value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $3.4 \times 10^{22}$  atoms/cm<sup>3</sup>, the minimum value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $3.1 \times 10^{22}$  atoms/cm<sup>3</sup>, and the average value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $3.3 \times 10^{22}$  atoms/cm<sup>3</sup>.

Further, as for the as-deposited Sample 3 (FIGS. 12A and 12B), the maximum value of the hydrogen concentration in the effective range of the measurement results of SIMS was 65 1.9×10<sup>20</sup> atoms/cm<sup>3</sup>, the minimum value of the hydrogen concentration in the effective range of the measurement

20

results of SIMS was  $1.1 \times 10^{20}$  atoms/cm<sup>3</sup>, and the average value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $1.4 \times 10^{20}$  atoms/cm<sup>3</sup>.

Note that, as for the  $N_2$ -baked Sample 3 (FIGS. 12A and 12B), the maximum value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $3.3 \times 10^{22}$  atoms/cm<sup>3</sup>, the minimum value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $3.2 \times 10^{22}$  atoms/cm<sup>3</sup>, and the average value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $3.2 \times 10^{22}$  atoms/cm<sup>3</sup>.

Further, as for the  $\rm N_2$ -baked Sample 3 (FIGS. 12A and 12B), the maximum value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $8.6 \times 10^{20}$  atoms/cm³, the minimum value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $1.4 \times 10^{20}$  atoms/cm³, and the average value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $2.7 \times 10^{20}$  atoms/cm³.

Note that, as for the air-baked Sample 3 (FIGS. 12A and 12B), the maximum value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $3.5 \times 10^{22}$  atoms/cm<sup>3</sup>, the minimum value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $3.3 \times 10^{22}$  atoms/cm<sup>3</sup>, and the average value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $3.4 \times 10^{22}$  atoms/cm<sup>3</sup>.

Further, as for the air-baked Sample 3 (FIGS. 12A and 12B), the maximum value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $1.0 \times 10^{21}$  atoms/cm<sup>3</sup>, the minimum value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $7.6 \times 10^{19}$  atoms/cm<sup>3</sup>, and the average value of the hydrogen concentration in the effective range of the measurement results of SIMS was  $2.8 \times 10^{20}$  atoms/cm<sup>3</sup>.

As described above, it is found that hydrogen is easily absorbed by the oxide semiconductor layer containing nitrogen.

In other words, an oxide semiconductor layer whose average value of the nitrogen concentration in the effective range of the measurement results of SIMS is  $1\times10^{20}$  atoms/cm<sup>3</sup> or less (less than  $1\times10^{20}$  atoms/cm<sup>3</sup>) is a dense layer which is difficult for hydrogen to enter.

In Samples 1 to 3, the nitrogen concentration in the oxide semiconductor layer was not increased even through the heat treatment. In Samples 1, in the case where the air-baking was performed, the nitrogen concentration in the oxide semiconductor layer was decreased.

Here, it is assumed that the reason why nitrogen is absorbed by the oxide semiconductor layer is that part of an oxygen bond in the oxide semiconductor layer is cut off during formation of the oxide semiconductor layer and a nitrogen bond is formed in a position where the oxide bond has been cut off.

Therefore, it is hypothesized that the concentration of nitrogen which is to be contained in the oxide semiconductor layer can be decreased in the case where an oxygen bond is formed in a position where another oxide bond has been cut off. Note that when these assumption and hypothesis are valid, oxide semiconductors other than an In—Ga—Zn—Obased oxide semiconductor are also influenced by nitrogen.

#### Example 2

In order to demonstrate the validity of the assumption and hypothesis in Example 1, an oxide semiconductor layer was formed using oxygen as a sputtering gas.

When oxygen is supplied, an oxygen bond should be more likely to be formed in a position where another oxygen bond has been cut off.

The oxide semiconductor layer was formed by sputtering using an oxide semiconductor target in which the atom ratio 5 of In to Ga and Zn was 1:1:0.5 ( $In_2O_3$ : $Ga_2O_3$ :ZnO=1:1:1 (at a molar ratio))

Note that before the formation of the oxide semiconductor layer, the reduction of leakage from a deposition chamber, the reduction of nitrogen on the inner wall of the deposition chamber, deposition onto a dummy substrate, or the like was sufficiently performed, so that incorporation of nitrogen into the oxide semiconductor layer was thoroughly prevented. (Sample 4)

In Sample 4, the flow rate of a sputtering gas was set as follows:  $Ar/O_2=30/15$  sccm (the proportion of  $O_2$  was

The measurement results of SIMS of Sample 4 are shown in FIG. 13A. (Sample 5)

In Sample 5 (the as-deposited sample), the flow rate of a sputtering gas was set as follows: Ar/O<sub>2</sub>=0/40 sccm (the proportion of  $O_2$  was 100%).

The measurement results of SIMS of Sample 5 are shown 25 in FIG. 13B.

(Consideration)

The effective range of the measurement results of SIMS is the same as that in Example 1.

Here, in Sample 4 (FIG. 13A), the maximum value of the 30 nitrogen concentration in the effective range of the measurement results of SIMS was  $5 \times 10^{19}$  atoms/cm<sup>3</sup> or less (less than  $5 \times 10^{19}$  atoms/cm<sup>3</sup>).

Therefore, in Sample 4 (FIG. 13A), the average value of the nitrogen concentration in the effective range of the measurement results of SIMS was also 5×10<sup>19</sup> atoms/cm<sup>3</sup> or less (less than 5×10<sup>19</sup> atoms/cm<sup>3</sup>) (the average value is not larger than the maximum value).

In Sample 4 (FIG. 13A), the average value of the nitrogen concentration in the effective range of the measurement 40 method using a silicon oxide target. results of SIMS was calculated. The average value was  $1\times10^{19}$  atoms/cm<sup>3</sup> or less (less than  $1\times10^{19}$  atoms/cm<sup>3</sup>).

In Sample 5 (FIG. 13B), the maximum value of the nitrogen concentration in the effective range of the measurement results of SIMS was  $2 \times 10^{19}$  atoms/cm<sup>3</sup> or less (less than 45  $2\times10^{19}$  atoms/cm<sup>3</sup>).

Therefore, in Sample 5 (FIG. 13B), the average value of the nitrogen concentration in the effective range of the measurement results of SIMS was also 1×10<sup>19</sup> atoms/cm<sup>3</sup> or less (less than  $1 \times 10^{19}$  atoms/cm<sup>3</sup>).

Sample 4 (FIG. 13A) and Sample 5 (FIG. 13B) have a lower nitrogen concentration than Sample 1 (FIG. 10B) to which oxygen was not added.

Note that, as for the as-deposited Sample 4 (FIG. 13A), the maximum value of the nitrogen concentration in the effective 55 range of the measurement results of SIMS was  $1.6 \times 10^{19}$ atoms/cm<sup>3</sup>, the minimum value of the nitrogen concentration in the effective range of the measurement results of SIMS was 2.2×10<sup>18</sup> atoms/cm<sup>3</sup>, and the average value of the nitrogen concentration in the effective range of the measurement 60 method. results of SIMS was 4.6×10<sup>18</sup> atoms/cm<sup>3</sup>.

Note that, as for the as-deposited Sample 5 (FIG. 13B), the maximum value of the nitrogen concentration in the effective range of the measurement results of SIMS was 1.6×10<sup>19</sup> atoms/cm<sup>3</sup>, the minimum value of the nitrogen concentration 65 in the effective range of the measurement results of SIMS was 2.3×10<sup>18</sup> atoms/cm<sup>3</sup>, and the average value of the nitrogen

22

concentration in the effective range of the measurement results of SIMS was 7.7×10<sup>18</sup> atoms/cm<sup>3</sup>.

Therefore, the validity of the assumption and hypothesis in Example 1 was demonstrated.

#### Example 3

The difference in electrical characteristics of transistors due to the difference in materials of the insulating layer 600 was examined.

(Common Condition)

The channel-etched transistor in FIG. 2B was manufac-

For the oxide semiconductor layer 410, a sample formed in the following manner was used: an oxide semiconductor layer was formed by sputtering using an oxide semiconductor target in which the atom ratio of In to Ga and Zn was 1:1:0.5 (In<sub>2</sub>O<sub>3</sub>:Ga<sub>2</sub>O<sub>3</sub>:ZnO=1:1:1 (at a molar ratio)), and the oxide semiconductor layer was etched into an island shape.

After the step of FIG. 1B, first heat treatment was performed at 350° C. for 1 hour in an air atmosphere.

After the step of FIG. 2B, a contact hole was formed in the insulating layer 600, and a wiring was formed over the insulating layer 600.

After the wiring was formed, second heat treatment was performed at 350° C. for 1 hour in an air atmosphere. (Sample 6)

The insulating layer 600 was formed by a sputtering method using a silicon oxide target.

The insulating layer 600 was formed under the following conditions. The substrate temperature was set to 100° C. The flow rate of a sputtering gas was set as follows: Ar/O<sub>2</sub>=40/10

The RF power source was used, and the power of the RF power source was set to 1.5 kW.

The deposition pressure was set to 0.4 Pa. (Sample 7)

The insulating layer 600 was formed by a sputtering

The insulating layer 600 was formed under the following conditions. The substrate temperature was set to 250° C. The flow rate of a sputtering gas was set as follows:  $Ar/O_2=40/10$ 

The RF power source was used, and the power of the RF power source was set to 1.5 kW.

The deposition pressure was set to 0.4 Pa. (Sample 8)

The insulating layer 600 was formed by a sputtering 50 method using a silicon oxide target.

The insulating layer 600 was formed under the following conditions. The substrate temperature was set to 100° C. The flow rate of a sputtering gas was set as follows: Ar/H<sub>2</sub>=46/4 sccm.

The RF power source was used, and the power of the RF power source was set to 1.5 kW.

The deposition pressure was set to 0.4 Pa. (Sample 9)

The insulating layer 600 was formed by a plasma CVD

The insulating layer 600 was formed under the following conditions. The substrate temperature was set to 200° C. The flow rate of deposition gases was set as follows: SiH<sub>4</sub>/  $N_2O=25/1000$  sccm.

(Sample 10)

The insulating layer 600 was formed by a plasma CVD method.

The insulating layer 600 was formed under the following conditions. The substrate temperature was set to 325° C. The flow rate of deposition gases was set as follows: SiH<sub>4</sub>/  $N_2O=27/1000$  secm.

(Sample 11)

The insulating layer 600 was formed using a siloxane film. (Sample 12)

The insulating layer 600 was formed using an acrylic film. (Sample 13)

The insulating layer 600 was formed using a polyimide 10 film.

(Consideration)

The threshold voltage Vth of Samples 8 to 13 shifted in a negative direction as compared with that of Samples 6 and 7.

substance which is intentionally made to contain a hydrogen element contains a hydrogen element.

Consequently, the insulating layer 600 containing hydrogen is in contact with a back channel.

The insulating layer 600 formed by a plasma CVD method 20 contains hydrogen and nitrogen.

Consequently, the insulating layer 600 containing hydrogen is in contact with the back channel.

The siloxane film, the acrylic film, and the polyimide film easily absorb moisture and easily release moisture.

Consequently, the insulating layer 600 which releases moisture is in contact with the back channel.

Here, with the use of computational science (it is also called simulation although there is a difference between computational science and simulation in a strict sense), calcula- 30 tion was performed to examine behavior in the case where a donor was present in the back channel. The calculation showed that, in the case where donor was present in the back channel, the threshold voltage shifted in a negative direction.

Therefore, it was found that, when the insulating layer 600 35 formed under the condition where hydrogen was not used as a sputtering gas was used, the threshold voltage was prevented from shifting in a negative direction.

In addition, it was found that oxygen was supplied to the back channel through the second heat treatment and oxygen 40 deficiency in the back channel was reduced in the case where the insulating layer 600 containing excessive oxygen which was formed using oxygen as a sputtering gas and using a silicon oxide target was used.

Note that in this example, the second heat treatment was 45 performed at 350° C. for 1 hour.

In the case where the insulating layer 600 was formed using an insulating layer containing excessive oxygen, Sample A on which the second heat treatment has not been performed and Sample B on which the second heat treatment has been per- 50 formed at 250° C. for 1 hour were prepared. Then, Sample A and Sample B were compared. The results showed that Sample B on which the second heat treatment has been performed at 250° C. for 1 hour had more excellent electrical characteristics of the transistor than Sample A.

Further, a transistor including the insulating layer 600 formed using only oxygen as a sputtering gas and using a silicon target was separately prepared.

The case where the insulating layer 600 was formed using only oxygen as a sputtering gas and using a silicon target at a 60 substrate temperature of 100° C. and the case where the insulating layer 600 was formed using only oxygen as a sputtering gas and using a silicon target at a substrate temperature of 200° C. were compared. The results showed that the case where the substrate temperature was set to 200° C. 65 was better than the case where the substrate temperature was set to 100° C., in the BT test.

24

The reason why the case where the substrate temperature was set to 200° C. was better in the BT test was found to be that, by increase in the substrate temperature during the formation of the insulating layer 600, moisture on a surface of the back channel was removed.

Therefore, the substrate temperature during the formation of the insulating layer 600 is preferably 200° C. or higher (the upper limit is not limited, but can be set to 300° C. or lower, 400° C. or lower, 500° C. or lower, 600° C. or lower, or 700° C. or lower).

#### Example 4

The channel-etched transistor in FIG. 2A (the In-Ga-Here, the insulating layer 600 formed with the use of a 15 Zn—O-based transistor) was manufactured. Note that the insulating layer 600 was not formed.

Then, a substrate over which the transistor has been formed was immersed in water.

The characteristics obtained before the substrate over which the transistor has been formed was immersed in water and the characteristics obtained just after the substrate over which the transistor has been formed was immersed in water were compared. The results showed that, just after the substrate over which the transistor has been formed was immersed in water, the threshold voltage shifted in a negative direction and off current was increased as compared with those before the substrate over which the transistor has been formed was immersed in water.

Next, the substrate which has been immersed in water was heated at 120° C. for 3 minutes in an air atmosphere to be dried, whereby the characteristics of the transistor were slightly recovered.

Next, the substrate which has been dried for 3 minutes was heated at 120° C. for 10 minutes (13 minutes in total) in an air atmosphere, whereby the characteristics of the transistor were further recovered.

Next, the substrate which has been dried for 13 minutes was heated at 120° C. for 40 minutes (53 minutes in total) in an air atmosphere, whereby the characteristics of the transistor became almost the same ones obtained before the substrate over which the transistor has been formed was immersed in water.

Thus, it is found that when moisture is attached to the surface of the back channel, the characteristics of the transistor are influenced.

A small amount of moisture such as the one which is attached to the surface of the back channel can be removed by heat treatment at 120° C. for 53 minutes or more.

Note that only the evaporation of moisture needs to be achieved, and it is therefore clear that increasing the temperature shortens the drying time.

#### Example 5

The effect of heat treatment performed on an oxide semiconductor layer was examined using thermal desorption spectroscopy (TDS).

Thermal desorption spectroscopy (TDS) is a method for analyzing a gas emitted when the temperature of a sample is increased.

In this example, emission of water vapor was examined. (Sample 14)

As Sample 14, a glass substrate was prepared. (Sample 15)

As Sample 15, a sample formed in the following manner was prepared: an oxide semiconductor layer was formed over a glass substrate by sputtering using an oxide semiconductor

target in which the atom ratio of In to Ga and Zn was 1:1:0.5  $(In_2O_3:Ga_2O_3:ZnO=1:1:1$  (at a molar ratio)).

Sample 15 has not been subjected to heat treatment (the as-deposited sample).

(Sample 16)

As Sample 16, the same sample as Sample 15 was prepared.

(Sample 17)

As Sample 17, a sample formed in the following manner was prepared: an oxide semiconductor layer was formed over 10 a glass substrate by sputtering using an oxide semiconductor target in which the atom ratio of In to Ga and Zn was 1:1:0.5 (In<sub>2</sub>O<sub>3</sub>:Ga<sub>2</sub>O<sub>3</sub>:ZnO=1:1:1 (at a molar ratio)).

After the oxide semiconductor layer was formed, heat treatment was performed at 250° C. for 1 hour in a nitrogen 15 atmosphere.

(Sample 18)

As Sample 18, a sample formed in the following manner was prepared: an oxide semiconductor layer was formed over a glass substrate by sputtering using an oxide semiconductor 20 target in which the atom ratio of In to Ga and Zn was 1:1:0.5 (In<sub>2</sub>O<sub>3</sub>:Ga<sub>2</sub>O<sub>3</sub>:ZnO=1:1:1 (at a molar ratio)).

After the oxide semiconductor layer was formed, heat treatment was performed at 350° C. for 1 hour in a nitrogen atmosphere.

(Sample 19)

As Sample 19, a sample formed in the following manner was prepared: an oxide semiconductor layer was formed over a glass substrate by sputtering using an oxide semiconductor target in which the atom ratio of In to Ga and Zn was 1:1:0.5 30 (In<sub>2</sub>O<sub>3</sub>:Ga<sub>2</sub>O<sub>3</sub>:ZnO=1:1:1 (at a molar ratio)).

After the oxide semiconductor layer was formed, heat treatment was performed at 450° C. for 1 hour in a nitrogen atmosphere.

(Consideration)

The results of the measurement using thermal desorption spectroscopy of Sample 14 and Sample 15 are shown in FIG. 14A.

Note that in FIG. **14**A, a graph **1414** corresponds to a graph of Sample 14 (only the glass substrate), and a graph **1415** 40 corresponds to a graph of Sample 15 (the glass substrate and the oxide semiconductor layer (the as-deposited sample)).

In FIG. 14A, there are a peak 1401 of moisture at around 50° C., a peak 1402 of moisture at around 100° C., and a peak 1403 of moisture at around 300° C.

As for Sample 14 including only the glass substrate, there are only the peak 1401 of moisture at around  $50^{\circ}$  C. and the peak 1402 of moisture at around  $100^{\circ}$  C., and there is not the peak 1403 of moisture at around  $300^{\circ}$  C.

On the other hand, as for Sample 15 including the oxide 50 semiconductor layer, there are the peak **1401** of moisture at around  $50^{\circ}$  C., the peak **1402** of moisture at around  $100^{\circ}$  C., and the peak **1403** of moisture at around  $300^{\circ}$  C.

Therefore, the peak **1403** of moisture at around 300° C. is a peak of moisture which is characteristic of the oxide semi- 55 conductor layer.

Thus, in the case where the peak 1403 of moisture at around  $300^{\circ}$  C. is detected, moisture should be contained in the oxide semiconductor layer.

The results of comparing Samples 16 to 19 are shown in 60 FIG. 14B.

Note that in FIG. 14B, a graph 1416 corresponds to a graph of Sample 16 (the as-deposited sample), a graph 1417 corresponds to a graph of Sample 17 (the sample heated at 250° C.), a graph 1418 corresponds to a graph of Sample 18 (the 65 sample heated at 350° C.), and a graph 1419 corresponds to a graph of Sample 19 (the sample heated at 450° C.).

26

As for Sample 17 (250 $^{\circ}$  C.) and Sample 18 (350 $^{\circ}$  C.), the peak **1403** of moisture at around 300 $^{\circ}$  C. was detected.

The number of the peaks 1403 of moisture at around  $300^\circ$  C. of Sample 17 (250 $^\circ$  C.) and Sample 18 (350 $^\circ$  C.) is smaller than that of Sample 16 (the as-deposited sample).

Thus, it was found that a certain amount of moisture was reduced through the heat treatment.

As for Sample 19 (450 $^{\circ}$  C.), there is not the peak **1403** of moisture at around 300 $^{\circ}$  C.

Therefore, it can be said that, in the case where heat treatment is performed at 450° C. for 1 hour or more, moisture is thoroughly eliminated. (Sample 20)

As Sample 20, a sample formed in the following manner was prepared: an oxide semiconductor layer was formed over a glass substrate by sputtering using an oxide semiconductor target in which the atom ratio of In to Ga and Zn was 1:1:0.5 (In<sub>2</sub>O<sub>3</sub>:Ga<sub>2</sub>O<sub>3</sub>:ZnO=1:1:1 (at a molar ratio)).

Then, Sample 20 was subjected to heat treatment at 650° C. for 3 minutes in a nitrogen atmosphere with the use of a gas RTA apparatus.

As for Sample 20 (650 $^{\circ}$  C.), there was not the peak of moisture at around 300 $^{\circ}$  C.

Thus, it can be said that, in the case where heat treatment is performed at 650° C. for 3 minutes or more, moisture is thoroughly eliminated.

Samples formed in the following manner were prepared: an oxide semiconductor layer was formed over a glass substrate by sputtering using an oxide semiconductor target in which the atom ratio of In to Ga and Zn was 1:1:1 (In $_2$ O $_3$ :Ga $_2$ O $_3$ : ZnO=1:1:2 (at a molar ratio)). Experiments similar to the above (comparison between the as-deposited sample, the sample heated at 250° C., the sample heated at 350° C., and the sample heated at 450° C.) were performed. The results showed that, as for only the sample on which the heat treatment was performed at 450° C. for 1 hour, there was not the peak of moisture at around 300° C.

Samples formed in the following manner were prepared: an oxide semiconductor layer was formed over a glass substrate by sputtering using an oxide semiconductor target in which the atom ratio of In to Ga and Zn was 1:1:4 ( $\rm In_2O_3$ : $\rm Ga_2O_3$ :  $\rm ZnO=1:1:8$  (at a molar ratio)). Experiments similar to the above (comparison between the as-deposited sample, the sample heated as 250° C., the sample heated at 350° C., and the sample heated at 450° C.) were performed. The results showed that, as for only the sample on which the heat treatment was performed at 450° C. for 1 hour, there was not the peak of moisture at around 300° C.

Therefore, it can be said that the oxide semiconductor layer which does not have a peak of moisture at around 300° C. in thermal desorption spectroscopy (the number of peaks is  $2\times10^{-11}$  or less at higher than or equal to  $100^{\circ}$  C. and lower than or equal to  $400^{\circ}$  C. (preferably, higher than or equal to  $250^{\circ}$  C. and lower than or equal to  $300^{\circ}$  C.) is an oxide semiconductor layer on which heat treatment has been performed at  $450^{\circ}$  C. or higher for 1 hour or more, or at  $650^{\circ}$  C. or higher for 3 minutes or more.

The absence of a peak of moisture at around  $300^{\circ}$  C. in thermal desorption spectroscopy means the absence of a peak of moisture at higher than or equal to  $100^{\circ}$  C. and lower than or equal to  $400^{\circ}$  C. in thermal desorption spectroscopy.

The absence of a peak of moisture at around 300° C. in thermal desorption spectroscopy means the absence of a peak of moisture due to the oxide semiconductor layer.

When the transistors having the structure of FIG. 2A were manufactured with only the conditions of first heat treatment changed, the transistors (250° C. and 350° C.) including an

oxide semiconductor layer having a peak of moisture at around 300° C. in thermal desorption spectroscopy had threshold voltage shifted in a negative direction and had significant deterioration in the BT test compared with the transistor (450° C.) including an oxide semiconductor layer 5 which did not have a peak of moisture at around 300° C. in thermal desorption spectroscopy.

Note that as the oxide semiconductor layer, an oxide semiconductor layer formed by sputtering using an oxide semiconductor target in which the atom ratio of In to Ga and Zn 10 was 1:1:0.5 ( $In_2O_3$ : $Ga_2O_3$ :ZnO=1:1:1 (at a molar ratio)) was used.

Note that before the formation of the oxide semiconductor layer, the reduction of leakage from a deposition chamber, the reduction of nitrogen on the inner wall of the deposition chamber, deposition onto a dummy substrate, or the like was sufficiently performed, so that an influence of nitrogen on the oxide semiconductor layer was thoroughly prevented.

Thus, the electrical characteristics of a transistor were found to be influenced by moisture in an oxide semiconductor 20 removed by heat treatment at least at 350° C. or lower. layer.

#### Example 6

measured by thermal desorption spectroscopy (TDS). (Sample 21)

As Sample 21, a sample formed in the following manner was prepared: an oxide semiconductor layer was formed over a glass substrate by sputtering using an oxide semiconductor 30 target in which the atom ratio of In to Ga and Zn was 1:1:0.5 (In<sub>2</sub>O<sub>3</sub>:Ga<sub>2</sub>O<sub>3</sub>:ZnO=1:1:1 (at a molar ratio)), and a silicon oxide film was formed over the oxide semiconductor layer by a sputtering method.

For the formation of the silicon oxide film, silicon oxide 35 was used as a target and the flow rate of gases was set as follows:  $Ar/O_2 = 40/10$  sccm.

Sample 21 has not been subjected to heat treatment. (Sample 22)

As Sample 22, a sample formed in the following manner 40 prevented. was prepared: an oxide semiconductor layer was formed over a glass substrate by sputtering using an oxide semiconductor target in which the atom ratio of In to Ga and Zn was 1:1:0.5 (In<sub>2</sub>O<sub>3</sub>:Ga<sub>2</sub>O<sub>3</sub>:ZnO=1:1:1 (at a molar ratio)), and a silicon oxide film was formed over the oxide semiconductor layer by 45 was thoroughly removed by heat treatment at 450° C. for 1 a sputtering method.

For the formation of the silicon oxide film, silicon oxide was used as a target and the flow rate of gases was set as follows: Ar/O<sub>2</sub>=40/10 sccm (oxygen was used as a sputtering

After that, heat treatment (in a furnace) was performed at 250° C. for 1 hour in a nitrogen atmosphere.

As Sample 23, a sample formed in the following manner was prepared: an oxide semiconductor layer was formed over 55 a glass substrate by sputtering using an oxide semiconductor target in which the atom ratio of In to Ga and Zn was 1:1:0.5 (In<sub>2</sub>O<sub>3</sub>:Ga<sub>2</sub>O<sub>3</sub>:ZnO=1:1:1 (at a molar ratio)), and a silicon oxide film was formed over the oxide semiconductor layer by a sputtering method.

For the formation of the silicon oxide film, silicon oxide was used as a target and the flow rate of gases was set as follows: Ar/O<sub>2</sub>=40/10 sccm (oxygen was used as a sputtering

After that, heat treatment (in a furnace) was performed at 65 350° C. for 1 hour in a nitrogen atmosphere. (Sample 24)

28

As Sample 24, a sample formed in the following manner was prepared: an oxide semiconductor layer was formed over a glass substrate by sputtering using an oxide semiconductor target in which the atom ratio of In to Ga and Zn was 1:1:0.5 (In<sub>2</sub>O<sub>3</sub>:Ga<sub>2</sub>O<sub>3</sub>:ZnO=1:1:1 (at a molar ratio)), and a silicon oxide film was formed over the oxide semiconductor layer by a sputtering method.

For the formation of the silicon oxide film, silicon oxide was used as a target and the flow rate of gases was set as follows: Ar/O<sub>2</sub>=40/10 sccm (oxygen was used as a sputtering

After that, heat treatment (in a furnace) was performed at 450° C. for 1 hour in a nitrogen atmosphere. (Consideration)

When the measurement was performed by thermal desorption spectroscopy (TDS), all the samples did not have a peak of moisture at around 300° C.

From the results of Example 5, moisture is not thoroughly

That is, Sample 23 and Sample 24 were supposed to have a peak of moisture at around 300° C.; however, a peak of moisture at around 300° C. was not detected.

The reason of this was found to be that the silicon oxide Degasification (moisture) of the following samples was 25 film formed using oxygen as a sputtering gas (i.e., the silicon oxide film containing excessive oxygen) blocked emission of moisture.

> Therefore, it was found that the silicon oxide film formed using oxygen as a sputtering gas (i.e., the silicon oxide film containing excessive oxygen) prevented diffusion of mois-

> In other words, it was found that the silicon oxide film containing excessive oxygen had an effect of blocking mois-

> Conversely, in the case where a silicon oxide film containing excessive oxygen is formed over an oxide semiconductor layer from which moisture has been released by first heat treatment, incorporation of moisture from the outside can be

#### Example 7

From the results of Example 5, it was found that moisture hour or more.

Thus, the hydrogen concentration in a sample which has been heated at a temperature higher than 450° C. was examined by secondary ion mass spectrometry (SIMS).

First, a plurality of samples in each of which an oxide semiconductor layer was formed over a glass substrate by sputtering using an oxide semiconductor target in which the atom ratio of In to Ga and Zn was 1:1:0.5 (In<sub>2</sub>O<sub>3</sub>:Ga<sub>2</sub>O<sub>3</sub>: ZnO=1:1:1 (at a molar ratio)) were prepared.

Further, a plurality of samples in each of which an oxide semiconductor layer was formed over a glass substrate by sputtering using an oxide semiconductor target in which the atom ratio of In to Ga and Zn was 1:1:1 (In<sub>2</sub>O<sub>3</sub>:Ga<sub>2</sub>O<sub>3</sub>: ZnO=1:1:2 (at a molar ratio)) were prepared.

Note that before the formation of the oxide semiconductor layer, the reduction of leakage from a deposition chamber, the reduction of nitrogen on the inner wall of the deposition chamber, deposition onto a dummy substrate, or the like was sufficiently performed, so that incorporation of nitrogen into the oxide semiconductor layer was thoroughly prevented.

The flow rate of a sputtering gas was set as follows:  $Ar/O_2=30/15$  sccm (the proportion of  $O_2$  was 33.3%).

sphere, and an atmosphere containing nitrogen and oxygen

Then, a plurality of samples which have been subjected to heat treatment at 450° C., 550° C., 600° C., and 650° C., (dry air (i.e., an atmosphere in which the ratio of nitrogen to respectively, were prepared. Note that the heating time for oxygen is 4:1)), respectively, were prepared. each sample was 1 hour.

Further, a plurality of samples which have been subjected 5 to heat treatment in a nitrogen atmosphere, an oxygen atmoThe results of comparison are shown in Table 1.

#### TABLE 1

		N2	O2	Dry Air
450° C. 1	hour [heat treatment]	_		
Sputtering apparatus A	In:Ga:Zn = 1:1:0.5 (atomic ratio) (In2O3:Ga2O3:ZnO =	$2 \times 10^{19} \text{ atoms/cm}^3$	$2 \times 10^{19} \text{ atoms/cm}^3$	$2 \times 10^{19} \text{ atoms/cm}^3$
	1:1:1 (molar ratio)) In:Ga:Zn = 1:1:1 (atomic ratio) (In2O3:Ga2O3:ZnO =	$3 \times 10^{19}$ atoms/cm <sup>3</sup>	$2 \times 10^{19} \text{ atoms/cm}^3$	$2 \times 10^{19}$ atoms/cm <sup>3</sup>
Sputtering apparatus B	1:1:2 (molar ratio)) In:Ga:Zn = 1:1:0.5 (atomic ratio) (In2O3:Ga2O3:ZnO =	$3 \times 10^{19} \text{ atoms/cm}^3$	$2 \times 10^{19} \text{ atoms/cm}^3$	$2 \times 10^{19}$ atoms/cm <sup>3</sup>
	1:1:1 (molar ratio)) In:Ga:Zn = 1:1:1 (atomic ratio) (In2O3:Ga2O3:ZnO = 1:1:2 (molar ratio))	$4 \times 10^{19} \text{ atoms/cm}^3$	$2 \times 10^{19} \text{ atoms/cm}^3$	$2 \times 10^{19} \text{ atoms/cm}^3$
550° C. 1	hour [heat treatment]	_		
Sputtering apparatus A	In:Ga:Zn = 1:1:0.5 (atomic ratio) (In2O3:Ga2O3:ZnO =	$3 \times 10^{18}$ atoms/cm <sup>3</sup>	$4 \times 10^{18}$ atoms/cm <sup>3</sup>	$4 \times 10^{18} \text{ atoms/cm}^3$
	1:1:1 (molar ratio)) In:Ga:Zn = 1:1:1 (atomic ratio) (In2O3:Ga2O3:ZnO = 1:1:2 (molar ratio))	$3 \times 10^{18} \text{ atoms/cm}^3$	$4 \times 10^{18} \text{ atoms/cm}^3$	$4 \times 10^{18} \text{ atoms/cm}^3$
Sputtering apparatus B	In:Ga:Zn = 1:1:0.5 (atomic ratio) (In2O3:Ga2O3:ZnO =	$3 \times 10^{18} \text{ atoms/cm}^3$	$4 \times 10^{18} \text{ atoms/cm}^3$	$4 \times 10^{18} \text{ atoms/cm}^3$
600° C. 1	1:1:1 (molar ratio)) In:Ga:Zn = 1:1:1 (atomic ratio) (In2O3:Ga2O3:ZnO = 1:1:2 (molar ratio)) I hour [heat treatment]	$3 \times 10^{18}$ atoms/cm <sup>3</sup>	$4 \times 10^{18}$ atoms/cm <sup>3</sup>	$4 \times 10^{18}$ atoms/cm <sup>3</sup>
Sputtering apparatus A	In:Ga:Zn = 1:1:0.5 (atomic ratio) (In2O3:Ga2O3:ZnO =	$6 \times 10^{18} \text{ atoms/cm}^3$	$4 \times 10^{18} \text{ atoms/cm}^3$	$4 \times 10^{18} \text{ atoms/cm}^3$
	1:1:1 (molar ratio)) In:Ga:Zn = 1:1:1 (atomic ratio) (In2O3:Ga2O3:ZnO =	$4 \times 10^{18} \text{ atoms/cm}^3$	$3 \times 10^{18} \text{ atoms/cm}^3$	$3 \times 10^{18} \text{ atoms/cm}^3$
Sputtering apparatus B	1:1:2 (molar ratio)) In:Ga:Zn = 1:1:0.5 (atomic ratio) (In2O3:Ga2O3:ZnO =	$4 \times 10^{18} \text{ atoms/cm}^3$	$5 \times 10^{18} \text{ atoms/cm}^3$	$4 \times 10^{18} \text{ atoms/cm}^3$
	1:1:1 (molar ratio)) In:Ga:Zn = 1:1:1 (atomic ratio) (In2O3:Ga2O3:ZnO = 1:1:2 (molar ratio))	$4 \times 10^{18}$ atoms/cm <sup>3</sup>	$4 \times 10^{18} \text{ atoms/cm}^3$	$4 \times 10^{18}$ atoms/cm <sup>3</sup>
650° C. 1	hour [heat treatment]	_		
Sputtering apparatus A	In:Ga:Zn = 1:1:0.5 (atomic ratio) (In2O3:Ga2O3:ZnO =	$4 \times 10^{18} \text{ atoms/cm}^3$	$4 \times 10^{18} \text{ atoms/cm}^3$	$4 \times 10^{18} \text{ atoms/cm}^3$
1:1:1 (molar ratio)) In:Ga:Zn = 1:1:1 (atomic ratio) (In2O3:Ga2O3:ZnO =	$4 \times 10^{18} \text{ atoms/cm}^3$	$4 \times 10^{18} \text{ atoms/cm}^3$	$4 \times 10^{18} \text{ atoms/cm}^3$	
Sputtering apparatus B	1:1:2 (molar ratio)) In:Ga:Zn = 1:1:0.5 (atomic ratio) (In2O3:Ga2O3:ZnO =	$4 \times 10^{18} \text{ atoms/cm}^3$	$4 \times 10^{18} \text{ atoms/cm}^3$	$4 \times 10^{18}$ atoms/cm <sup>3</sup>
	1:1:1 (molar ratio)) In:Ga:Zn =	$4 \times 10^{18} \text{ atoms/cm}^3$	$4 \times 10^{18} \text{ atoms/cm}^3$	$4 \times 10^{18} \text{ atoms/cm}^3$

**30** 

TABLE 1-continued

	N2	O2	Dry Air
1:1:1 (atomic ratio) (In2O3:Ga2O3:Zn0 1:1:2 (molar ratio))	) =		

In Table 1, the values measured by secondary ion mass spectrometry (SIMS) represent the average values in the  $^{10}$  effective range of SIMS.

From Table 1, it was found that, in the case where heat treatment was performed at 550° C. or higher for 1 hour or more, the hydrogen concentration in the oxide semiconductor layer was significantly decreased.

In other words, the oxide semiconductor layer having a hydrogen concentration of  $6\times10^{18}$  atoms/cm<sup>3</sup> or less is an oxide semiconductor layer which has been subjected to heat treatment at high temperature (at  $550^{\circ}$  C. or higher for 1 hour or more).

Note that when the transistors having the structure shown in FIG. 2A were manufactured with only the conditions of first heat treatment changed, the transistor ( $450^{\circ}$  C.) including an oxide semiconductor layer having a hydrogen concentration of  $1\times10^{19}$  atoms/cm<sup>3</sup> or more had threshold voltage 25 shifted in a negative direction and had significant deterioration in the BT test compared with the transistors ( $550^{\circ}$  C.,  $600^{\circ}$  C., and  $650^{\circ}$  C.) each including an oxide semiconductor layer having a hydrogen concentration of less than  $1\times10^{19}$  atoms/cm<sup>3</sup>.

Note that as the oxide semiconductor layer, an oxide semiconductor layer formed by sputtering using an oxide semiconductor target in which the atom ratio of In to Ga and Zn was 1:1:0.5 ( $\ln_2 O_3$ : $Ga_2 O_3$ :ZnO=1:1:1 (at a molar ratio)) was used.

Note that before the formation of the oxide semiconductor layer, the reduction of leakage from a deposition chamber, the reduction of nitrogen on the inner wall of the deposition chamber, deposition onto a dummy substrate, or the like was sufficiently performed, so that an influence of nitrogen on the 40 oxide semiconductor layer was thoroughly prevented.

According to the above, the electrical characteristics of a transistor were found to be influenced by hydrogen in an oxide semiconductor layer.

#### Example 8

FIG. 16 is a photograph of a cross section of a thin film transistor including an In—Ga—Zn—O-based oxide semi-conductor which is observed with a transmission electron 50 microscope (TEM, H-9000-NAR manufactured by Hitachi, Ltd., 300 kV).

The thin film transistor shown in FIG. 16 is a sample obtained in such a manner that an In—Ga—Zn—O-based oxide semiconductor layer having a thickness of 50 nm was 55 formed as an oxide semiconductor layer 1601, first heat treatment (at 650° C. for 1 hour) was performed in a nitride atmosphere, a titanium layer 1604 having a thickness of 150 nm was formed as a metal film, and further, second heat treatment (at 250° C. for 1 hour) was performed in a nitrogen 60 atmosphere.

In FIG. 16, at the interface between the oxide semiconductor layer 1601 and the titanium layer 1604, an indium-rich layer 1602 and a titanium oxide layer 1603 can be detected.

Note that the indium-rich layer **1602** and the titanium oxide 65 layer **1603** were detected by a fast fourier transform mapping (FFTM) method.

It was found that, through the second heat treatment in the state where the titanium layer 1604 and the oxide semiconductor layer 1601 were in contact with each other, oxygen was extracted from the oxide semiconductor layer and the titanium oxide layer 1603 was formed.

Further, it was found that a portion where oxygen was extracted in the oxide semiconductor layer 1601 became the indium-rich layer 1602 in which a crystal of indium was precipitated.

In this manner, through the second heat treatment in the state where the titanium layer 1604 and the oxide semiconductor layer 1601 are in contact with each other, the titanium oxide layer 1603 can be formed.

Note that it is clear that a similar reaction occurs even when an oxide semiconductor layer other than an In—Ga—Zn—O-based oxide semiconductor layer is used because a reaction of this example is a reaction between oxygen and titanium (Ti).

Therefore, the second heat treatment is preferably performed at  $250^{\circ}$  C. or higher (the upper limit is not limited, but can be set to  $300^{\circ}$  C. or lower,  $400^{\circ}$  C. or lower,  $500^{\circ}$  C. or lower, or  $700^{\circ}$  C. or lower).

Note that a material having low resistance (a film containing aluminum as its main component, a film containing copper as its main component, or the like) is preferably formed over the titanium layer 1604 because the wiring resistance can be reduced.

This application is based on Japanese Patent Application serial no. 2009-281505 filed with Japan Patent Office on Dec. 11, 2009, the entire contents of which are hereby incorporated by reference.

The invention claimed is:

 A method for manufacturing a semiconductor device comprising a gate electrode, an oxide semiconductor layer, a gate insulating layer provided between the gate electrode and
 the oxide semiconductor layer, and an insulating layer in contact with the oxide semiconductor layer, the method comprising the steps of:

performing a first heat treatment on the oxide semiconductor layer having a nitrogen concentration of  $1\times10^{20}$  atoms/cm³ or less at 350° C. or higher for 1 hour or more, and

performing a second heat treatment on the oxide semiconductor layer so that oxygen in the insulating layer is supplied to the oxide semiconductor layer.

- 2. The method for manufacturing a semiconductor device according to claim 1, wherein the oxide semiconductor layer comprises indium, gallium, and zinc.
- 3. The method for manufacturing a semiconductor device according to claim 1, wherein the first heat treatment is performed under an atmosphere containing nitrogen.
- 4. The method for manufacturing a semiconductor device according to claim 1, wherein the second heat treatment is performed at higher than or equal to  $150^{\circ}$  C. and lower than or equal to  $500^{\circ}$  C.
- 5. A method for manufacturing a semiconductor device comprising a gate electrode, an oxide semiconductor layer, a gate insulating layer provided between the gate electrode and

the oxide semiconductor layer, and an insulating layer in contact with the oxide semiconductor layer, the method comprising the steps of:

performing a first heat treatment on the oxide semiconductor layer having a nitrogen concentration of  $1\times10^{20}$  atoms/cm<sup>3</sup> or less at 450° C. or higher for 1 hour or more, and

performing a second heat treatment on the oxide semiconductor layer so that oxygen in the insulating layer is supplied to the oxide semiconductor layer.

- **6**. The method for manufacturing a semiconductor device according to claim **5**, wherein the oxide semiconductor layer comprises indium, gallium, and zinc.
- 7. The method for manufacturing a semiconductor device according to claim 5, wherein the first heat treatment is performed under an atmosphere containing nitrogen.
- **8**. The method for manufacturing a semiconductor device according to claim **5**, wherein the second heat treatment is performed at higher than or equal to  $150^{\circ}$  C. and lower than or equal to  $500^{\circ}$  C.
- 9. A method for manufacturing a semiconductor device comprising a gate electrode, an oxide semiconductor layer, a gate insulating layer provided between the gate electrode and the oxide semiconductor layer, and an insulating layer in contact with the oxide semiconductor layer, the method comprising the steps of:

performing a first heat treatment on the oxide semiconductor layer having a nitrogen concentration of  $1\times10^{20}$  atoms/cm $^3$  or less at 550° C. or higher for 1 hour or more, and

performing a second heat treatment on the oxide semiconductor layer so that oxygen in the insulating layer is supplied to the oxide semiconductor layer.

34

- 10. The method for manufacturing a semiconductor device according to claim 9, wherein the oxide semiconductor layer comprises indium, gallium, and zinc.
- 11. The method for manufacturing a semiconductor device according to claim 9, wherein the first heat treatment is performed under an atmosphere containing nitrogen.
- 12. The method for manufacturing a semiconductor device according to claim 9, wherein the second heat treatment is performed at higher than or equal to  $150^{\circ}$  C. and lower than or equal to  $500^{\circ}$  C.
- 13. A method for manufacturing a semiconductor device comprising a gate electrode, an oxide semiconductor layer, a gate insulating layer provided between the gate electrode and the oxide semiconductor layer, and an insulating layer in contact with the oxide semiconductor layer, the method comprising the steps of:

performing a first heat treatment on the oxide semiconductor layer having a nitrogen concentration of  $1\times10^{20}$  atoms/cm³ or less at 650° C. or higher for 3 minutes or more, and

performing a second heat treatment on the oxide semiconductor layer so that oxygen in the insulating layer is supplied to the oxide semiconductor layer.

- 14. The method for manufacturing a semiconductor device according to claim 13, wherein the oxide semiconductor layer comprises indium, gallium, and zinc.
- 15. The method for manufacturing a semiconductor device according to claim 13, wherein the first heat treatment is performed under an atmosphere containing nitrogen.
- 16. The method for manufacturing a semiconductor device according to claim 13, wherein the second heat treatment is performed at higher than or equal to  $150^{\circ}$  C. and lower than or equal to  $500^{\circ}$  C.

\* \* \* \* \*